

Pozytony - narzędzie badania strukturalnych

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Positron identity

e^+ is antiparticle of e^- :

- mass $511.003 \text{ keV}/c^2$
- spin $\frac{1}{2}$
- opposite Q
- opposite μ_B
- stable in vacuum ($>2 \times 10^{21} \text{ y}$)

Ps is light H :

- Energy $E = \frac{1}{2} \text{ Ry}$
- p-Ps: $\tau = 125 \text{ ps}, 2\gamma$
- o-Ps: $\tau = 142 \text{ ns}, 3\gamma$



One of Anderson's (1933) original photographs illustrating the historic discovery of the positron. In the cloud chamber, there is a lead plate 6 mm thick and a magnetic field oriented in the page. The change of energy (63 MeV below the plate to 23 MeV above) with the known thickness of lead and magnitude of the field proves that the particle is positive and of the same mass as the electron.

Positron history

History of “slow” positrons

1930 – e^+ postulated by Dirac

1932 – discovered in cosmic rays
by Anderson

“out of 1300 photographs of cosmic tracks,
15 were od positive particles which could not
have a mass greater as that of the proton”

1950 – Madanski-Rasetti try to moderate

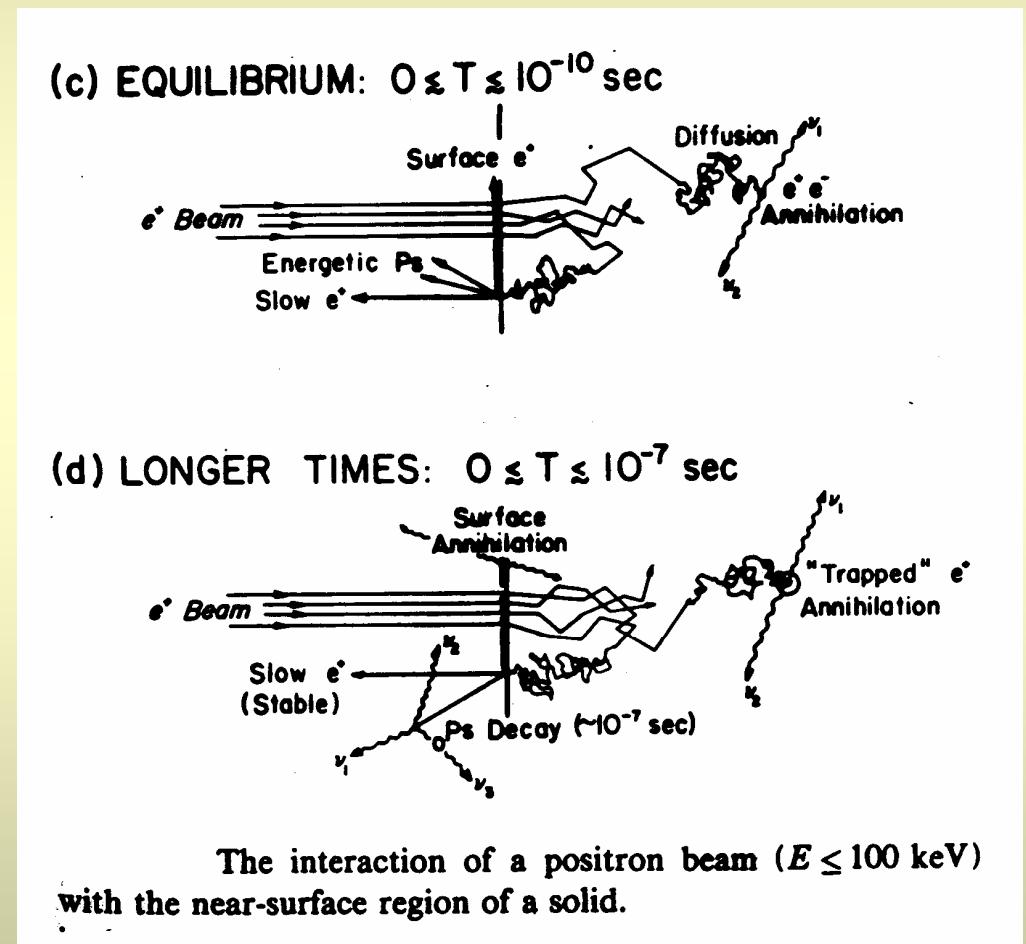
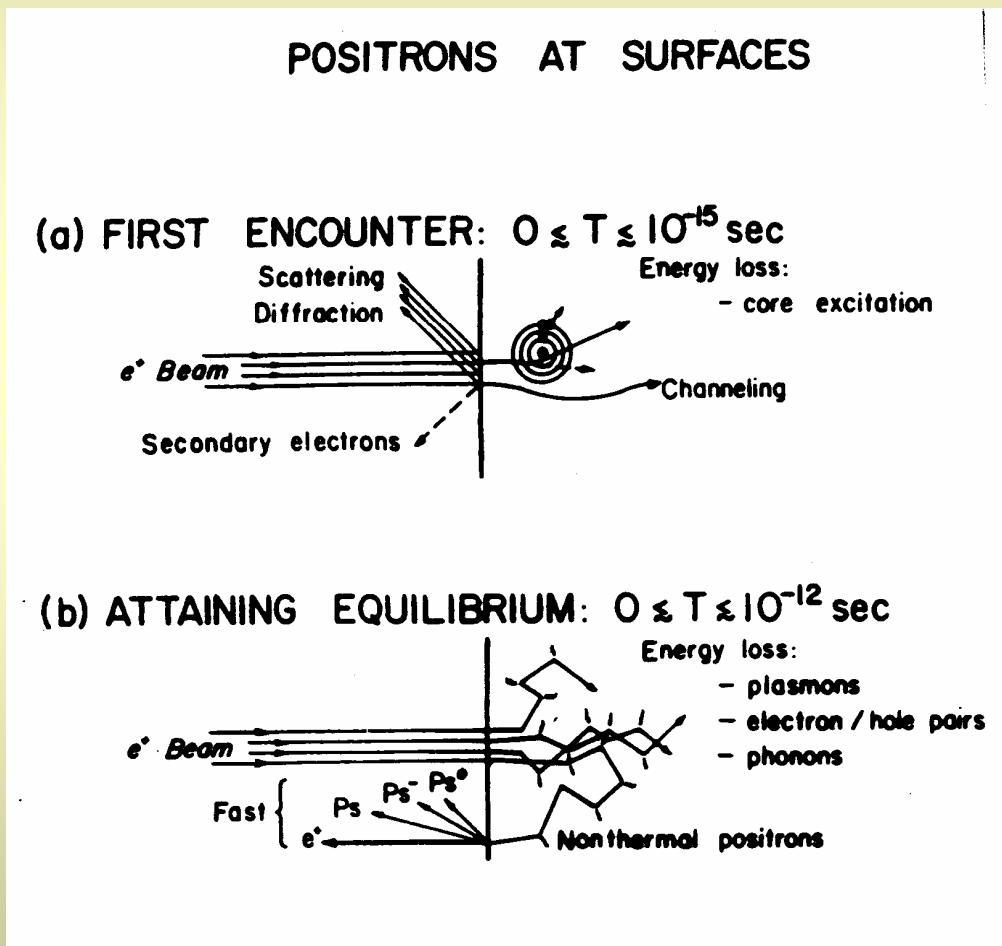
1951 – evidence of Ps atom (Deutsch)

1958 – moderated e^+ , $\varepsilon=3 \times 10^{-8}$ (Cherry)

1979 – single crystal moderator (Mills)

1980 – brightness enhancement (Mills)

Positron slowing down



Positron sources

Radioactive nuclides

Nuclide	Half-life	E [MeV]	f [%]	E_γ [MeV]
¹¹ C	20.4min	0.96	100	-
¹⁸ F	110min	0.633	97	-
²² Na	2.6y	0.546	90	1.275
⁵⁸ Co	70.8d	0.475	15	0.811
⁶⁴ Cu	12.7h	0.653	19	-
⁸⁸ Ge	271d	1.90	90	-

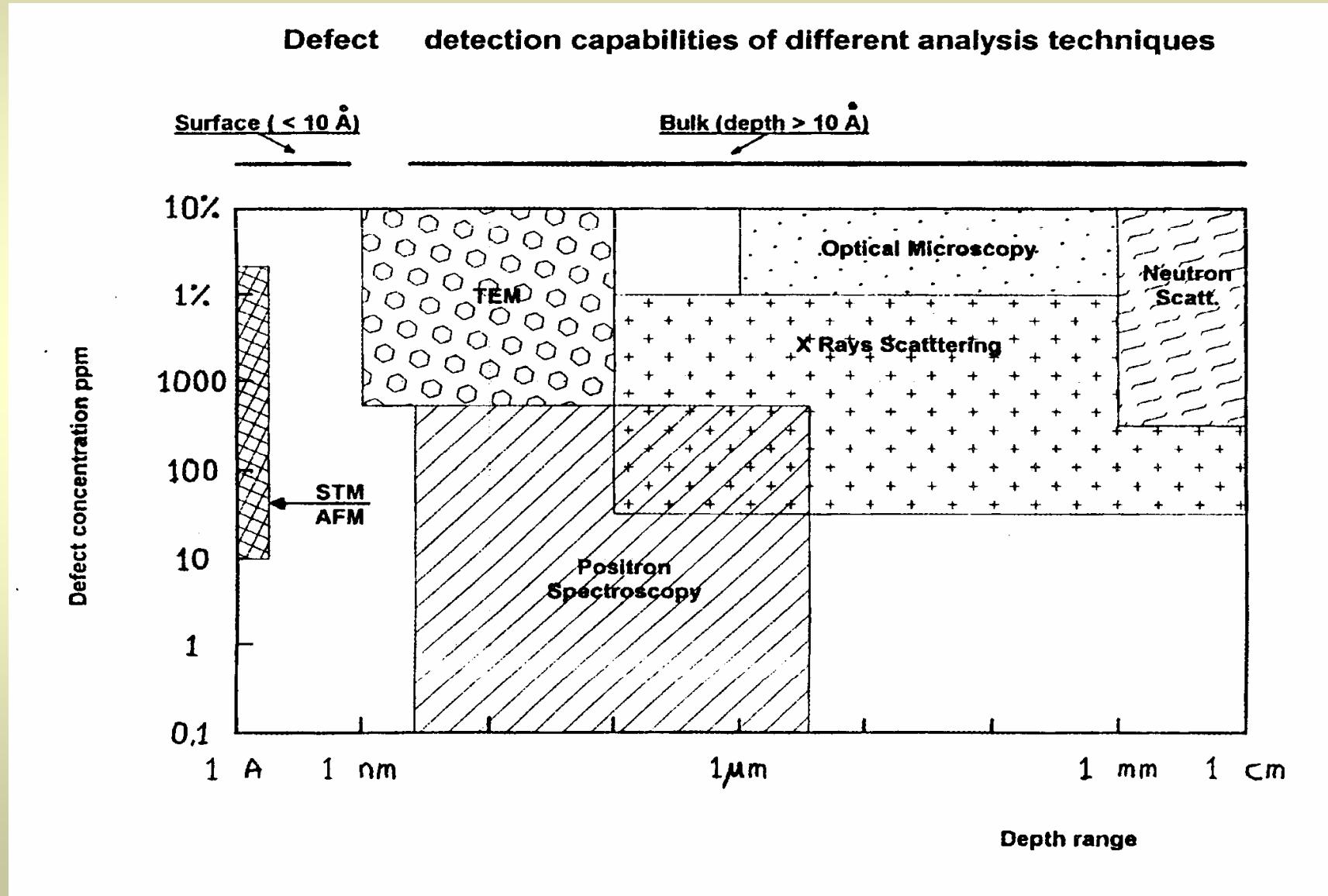
Moderators

Element	W^- (eV)	W^+ (eV)	W^+ (th)
C(001)	3.7	-3.03	
C(0001)		1.4	
Si(111)	4.74	? 0	2.21
Ni(100)	5.22	-1.0	-0.77
W(100)	5.25	-3.0	
Au	5.2	0.9	1.1

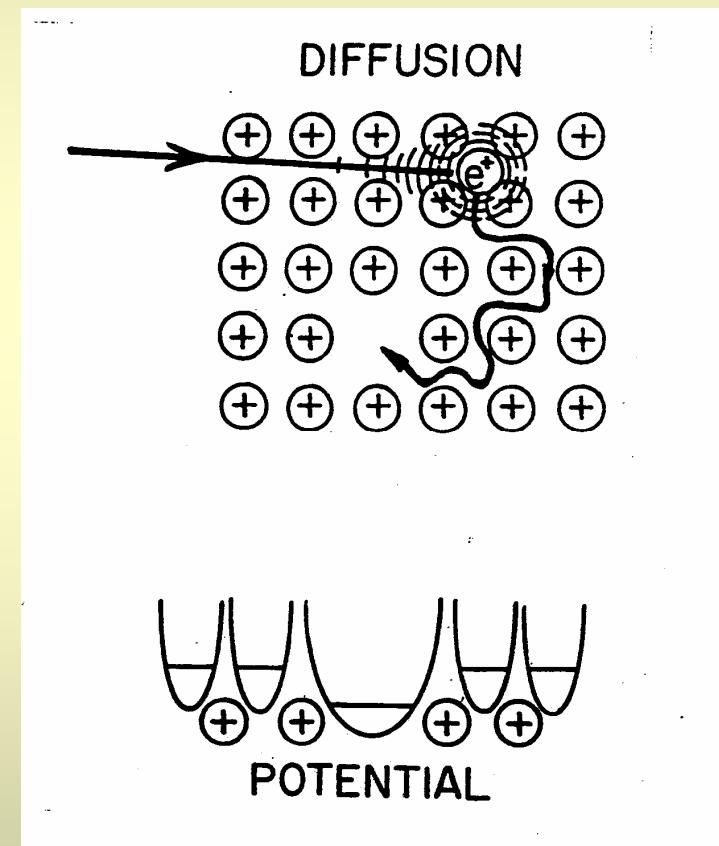
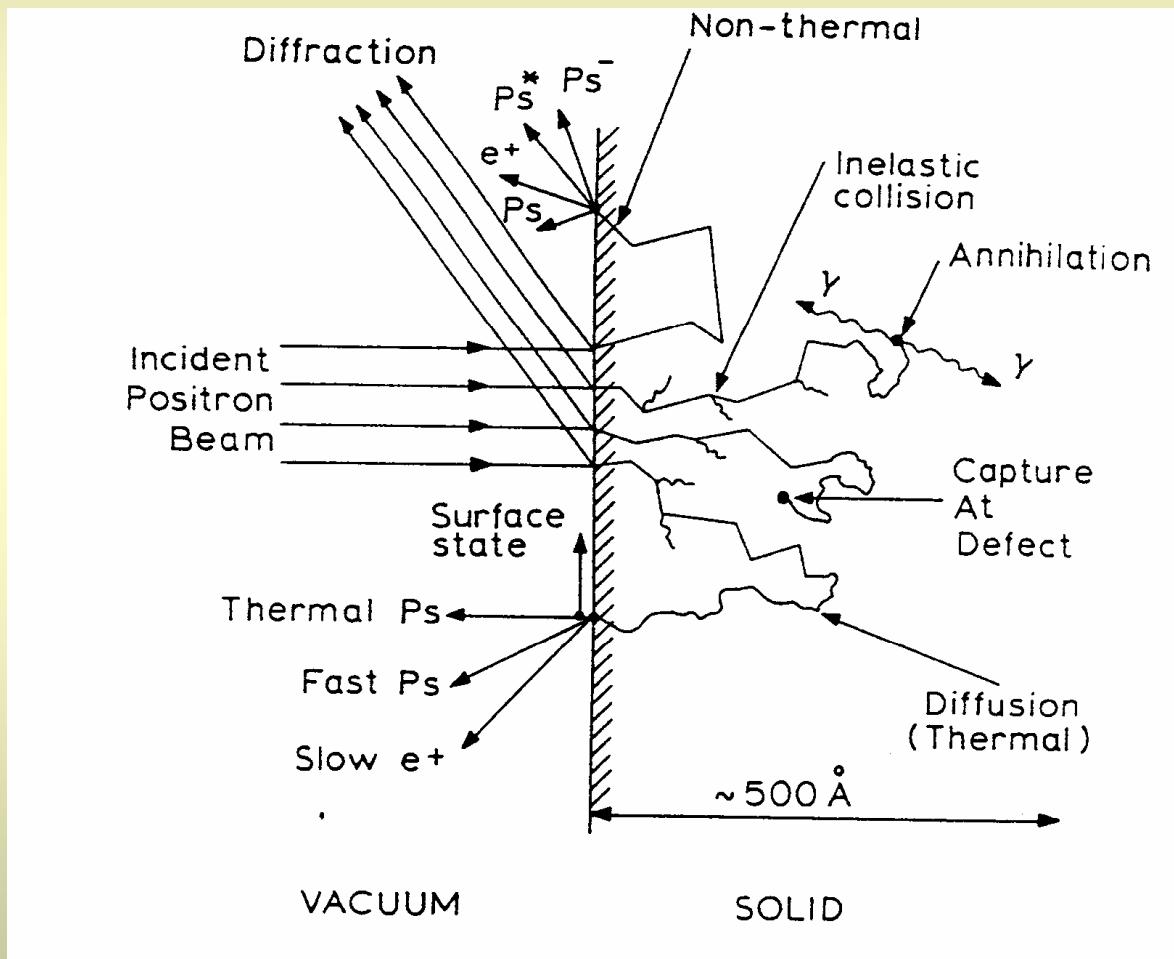
W (100): $\varepsilon = 4 \times 10^{-4}$

Solid Ne: $\varepsilon = 1\%$?

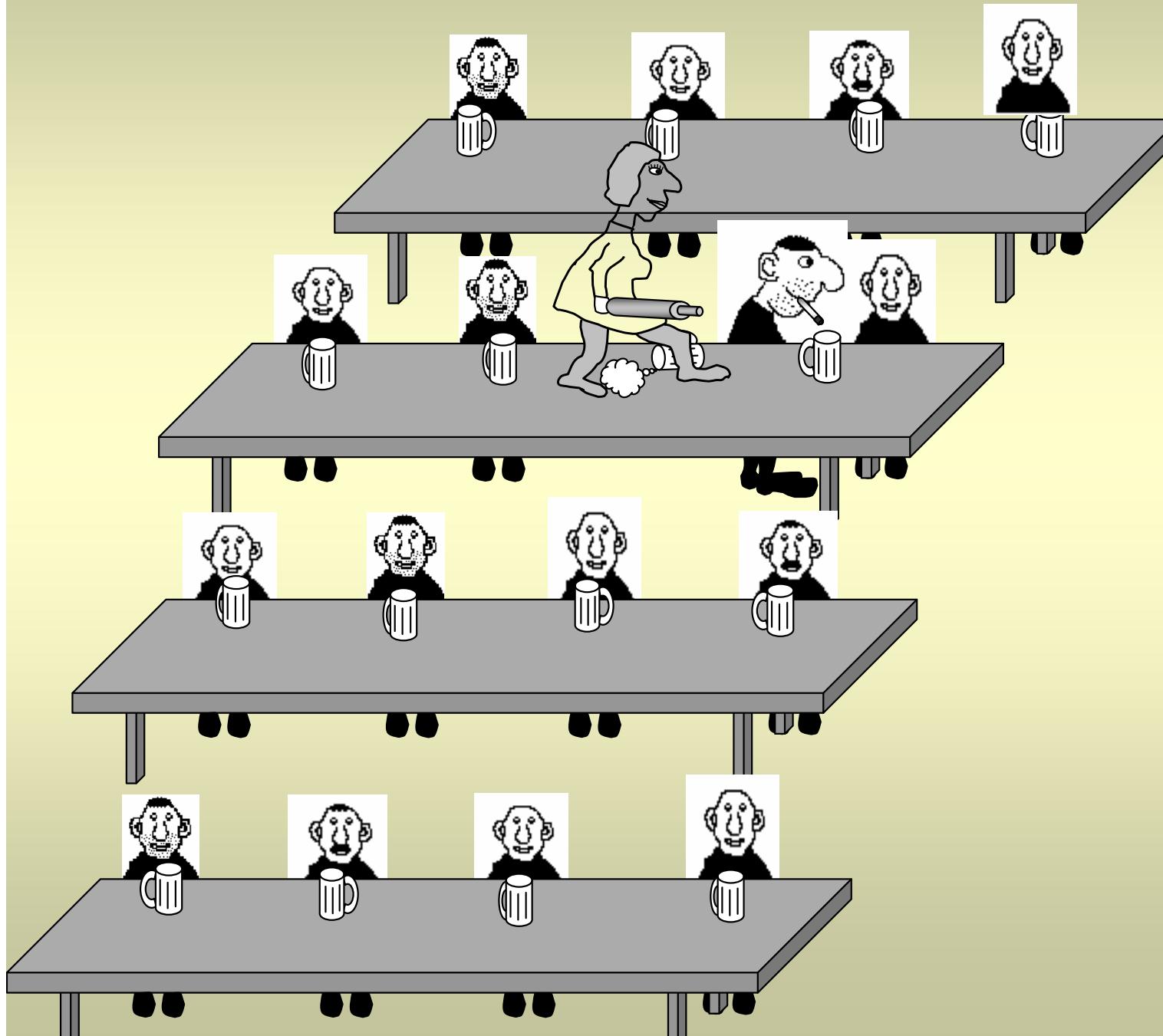
Positrons in Solid State Physics



Positron walking



Positron in a crystal



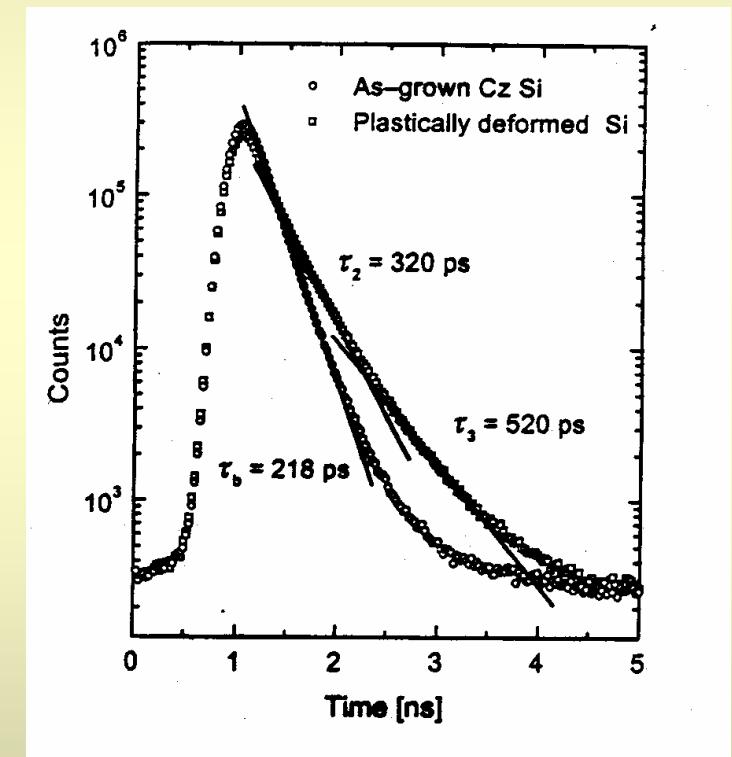
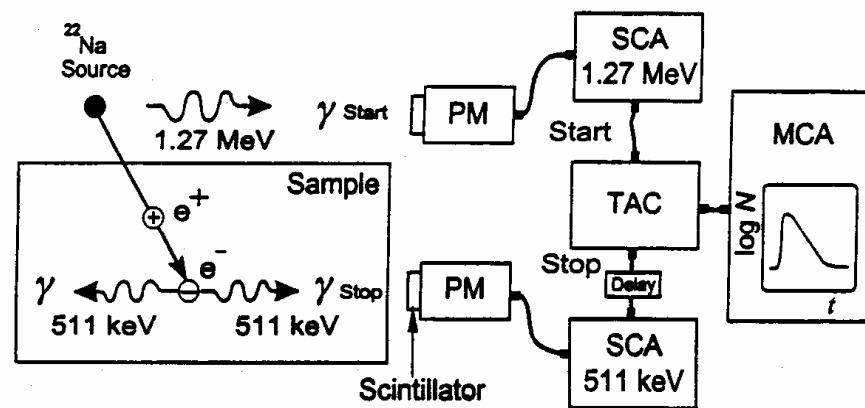
Techniques:

- Doppler broadening (depth profile)
- lifetime (in bulk)
- coincidence (in bulk)

Samples:

- semiconductors
- metal-doped glasses
- pure metals and alloys

1. Positron lifetime technique



$$\tau_{\text{defect}} > \tau_{\text{bulk}}$$

1. Life-time technique

High-purity metals (monocrystals)

Z	Element	τ_{exp}	$\tau_{\text{th}}[9]$	$\tau_{\text{th}}[20]$
5	Be	209	-	
13	Al	160	166	163.3
14	Si	220	221	217.0
22	Ti	145	146	
23	V	123	116	129.8
28	Ni	105	96	112.1
29	Cu	115	106	122.8
32	Ge	227	228	
40	Zr	159	159	
74	W	115	100	
79	Au	206	187	120.3

[20] A. Rubaszek, Z. Szotek, W. M. Temmerman, Phys. Rev. B 58 (1998) 11285

1. Life-time technique

High-purity metals (annealing)

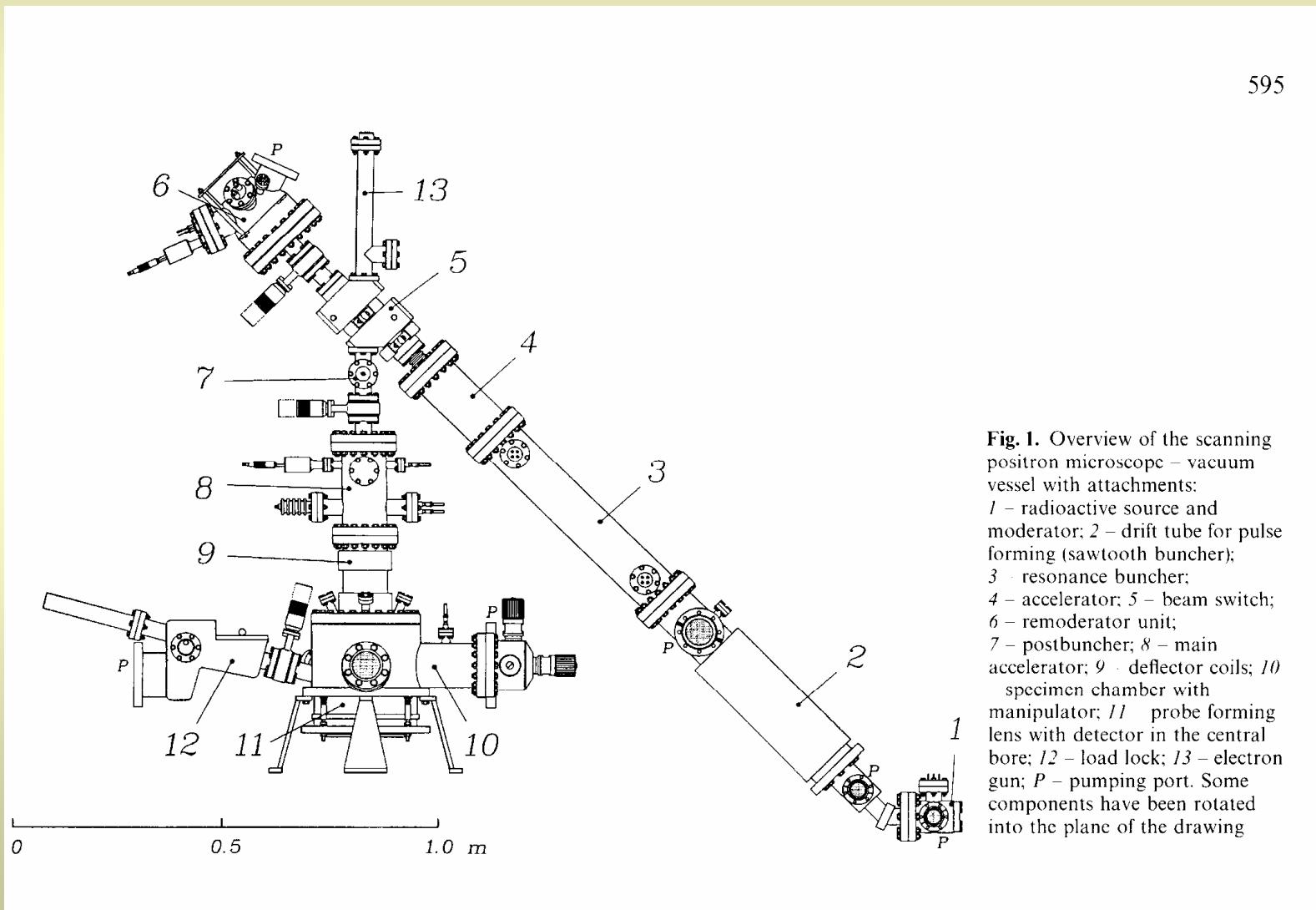
Z	Elem- ent ^{a)}	Purity	Temp. °C ^{b)}	τ_m [ps]	τ_1 ^{c)} [ps]	τ_2 [ps]	I_2 ^{d)} [%]	Theory [20]
28	Ni	99.98	1000	105	-	-	-	96
29	Cu	99.8	850	115	-	-	-	106
40	Zr	99.8	1100	159	-	-	-	159
42	Mo	99.9	> 1500	119	-	-	-	111
				123	117.4	660±60	1.0	
47	Ag	99.9	850	135	-	-	-	120
				188	122.7	240±2	57.0	
74	W	99.8	>2000	115	103	245±15	8.6	100
78	Pt	99.5	1000	167	142	282±8	18.0	94
79	Au	99.99	850	123	119	230±56	3.7	107
				190	181	290±40	7.9	
82	Pb	99.5	250	206	-	-	-	187
				210	201.3	960±75	1.2	

Trento-München Positron Microscope

E=500 eV – 25 keV

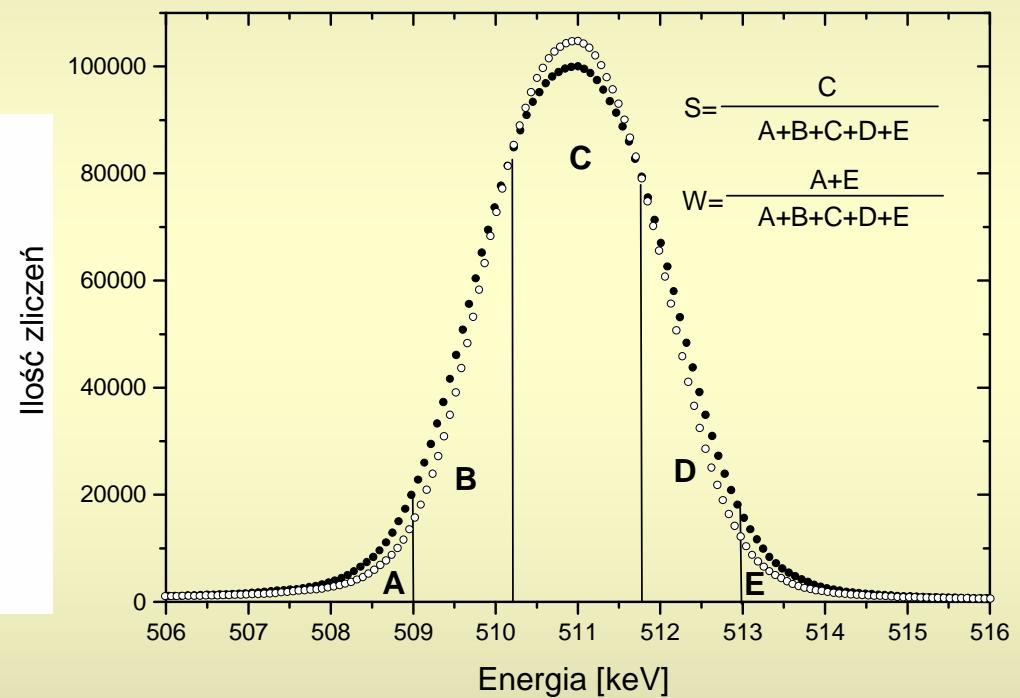
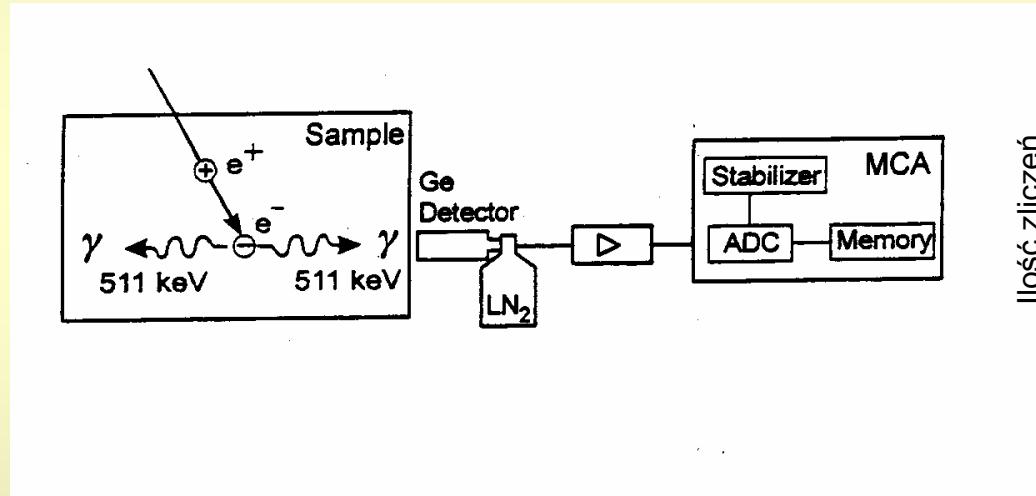
spot = 2 μm

595



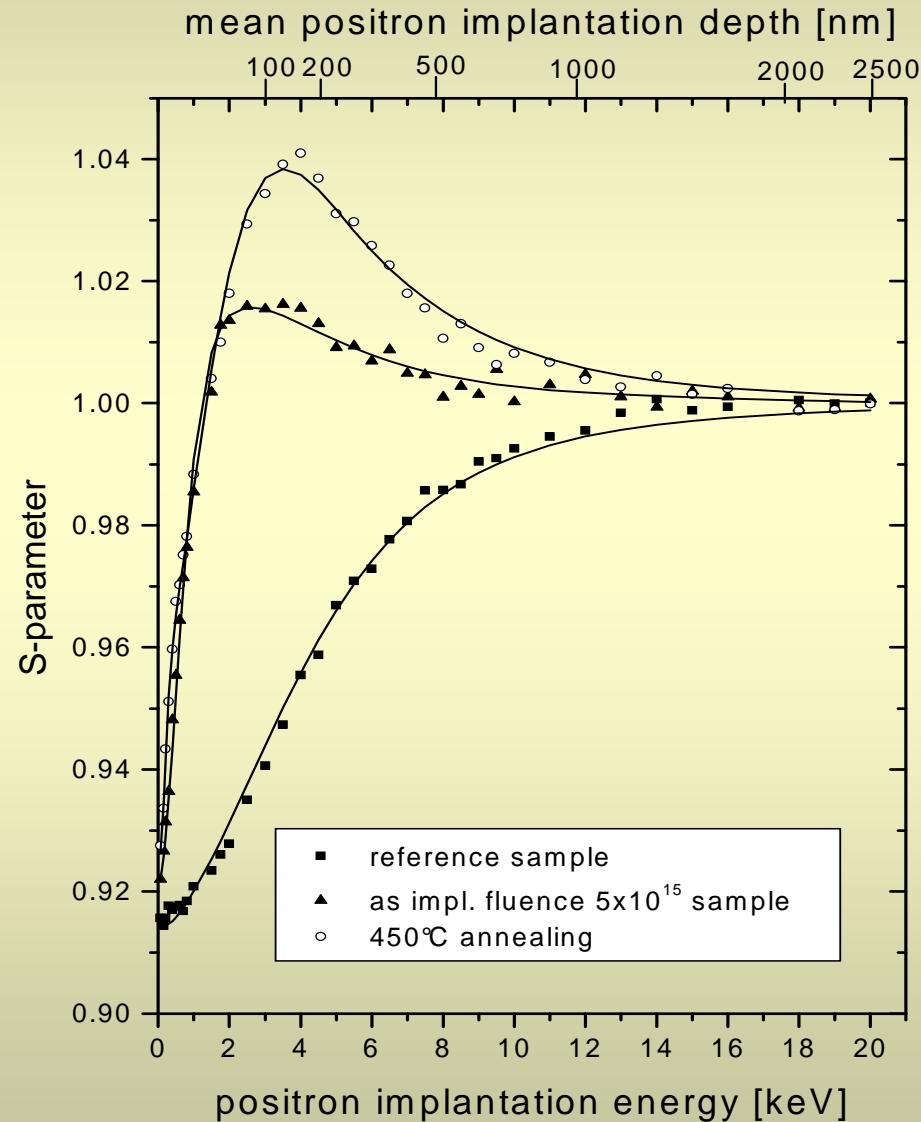
2. Doppler broadening technique

$$\mathbf{p}_{\text{tot}} = \mathbf{p}_e + \mathbf{p}_p \quad \Delta E = c p_z / 2$$

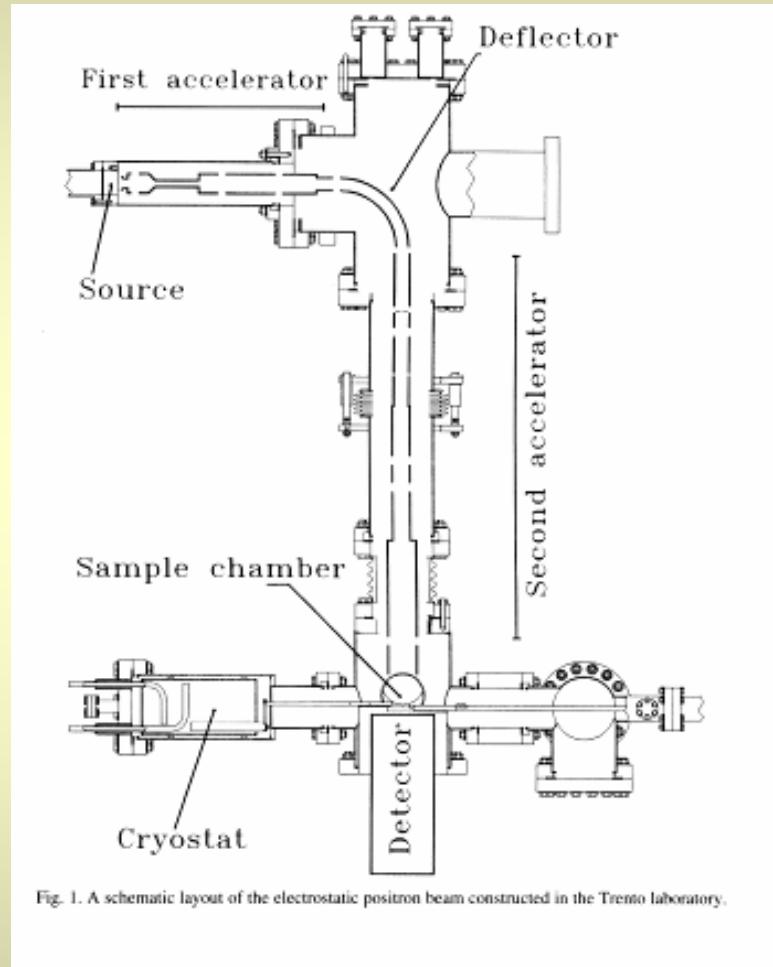


$$S = (E_0 \pm 0.85 \text{ keV}) / (E_0 \pm 4.25 \text{ keV})$$

2. Doppler-broadening: normalization



Trento Positron Annihilation Set-up



$E=100 \text{ eV} - 25 \text{ keV}$ (down to $2\mu\text{m}$)
spot < 1 mm

He bubbles in Si

He – implantation

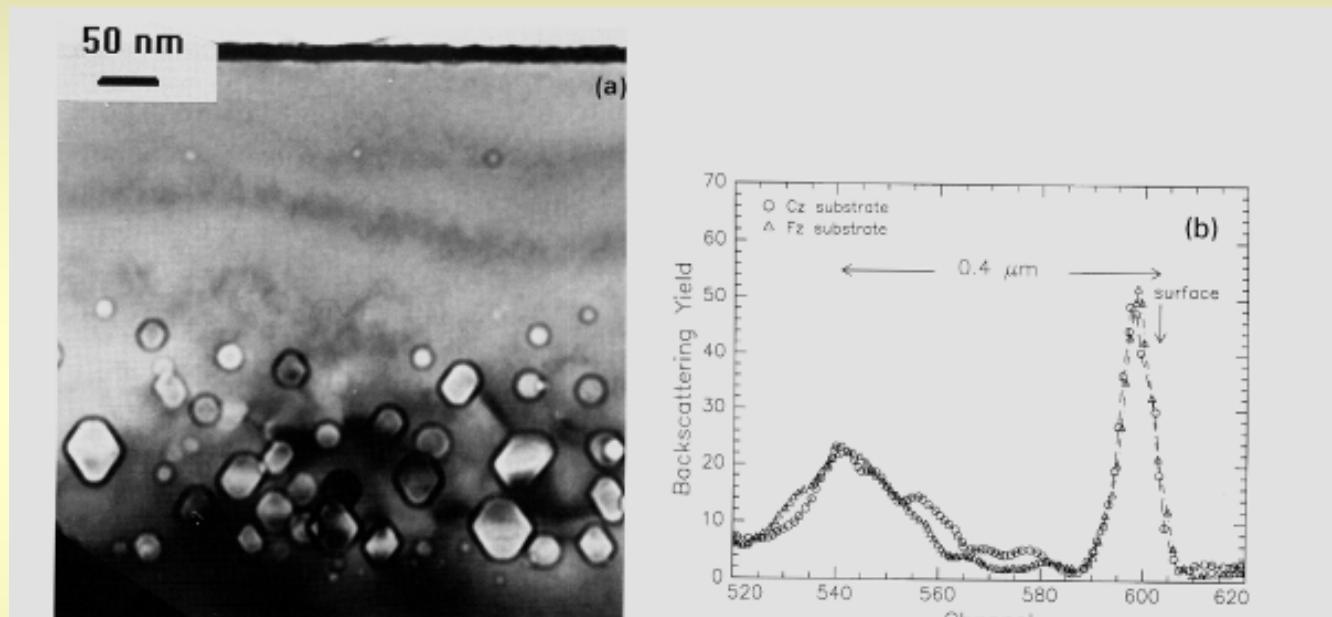
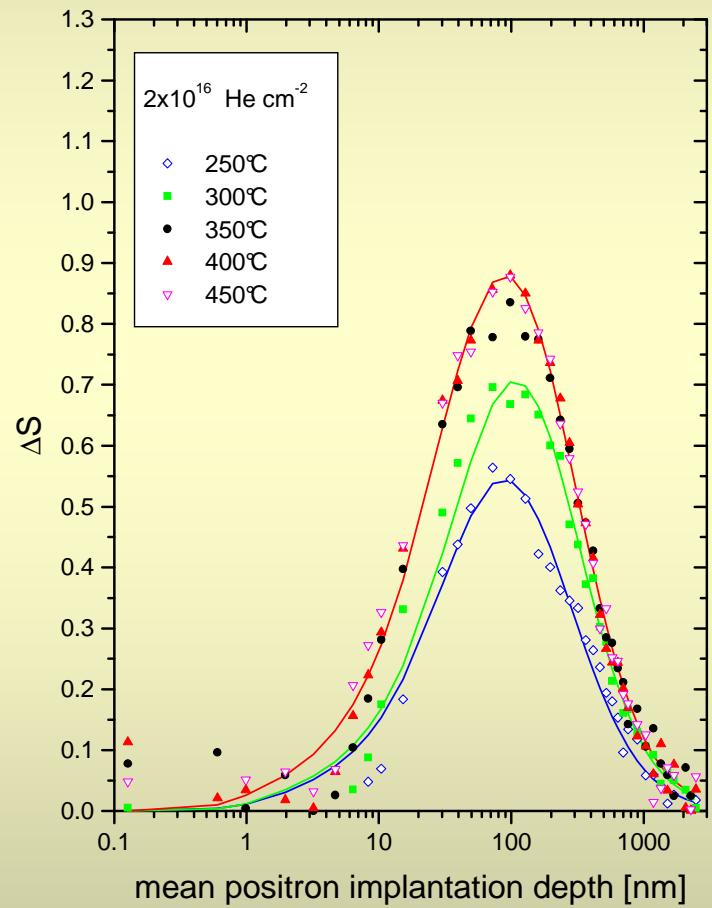
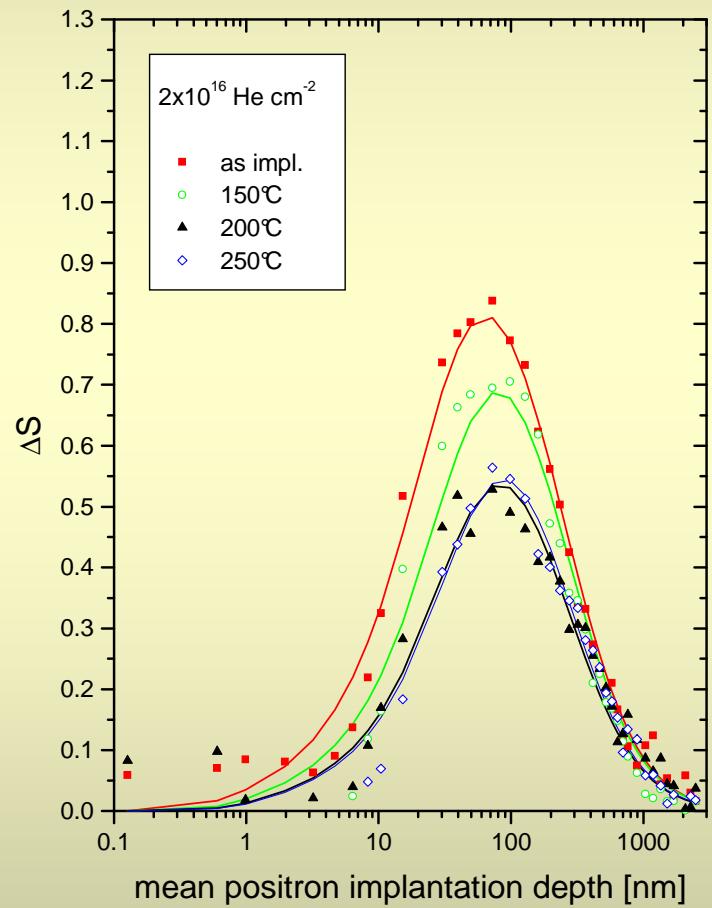


Fig. 3. (a) XTEM image of a double polished sample implanted with $1 \times 10^{17}/\text{cm}^2$ He at 40 keV and thermal processed at 700°C for 1 h. Then it was implanted in the opposite side with $5 \times 10^{14}/\text{cm}^2$ Pt at 1 MeV and diffused at 1000°C for 2 h. In this case the He implanted side is shown. (b) RBS spectra of the same sample as in (a) of the He implanted side.

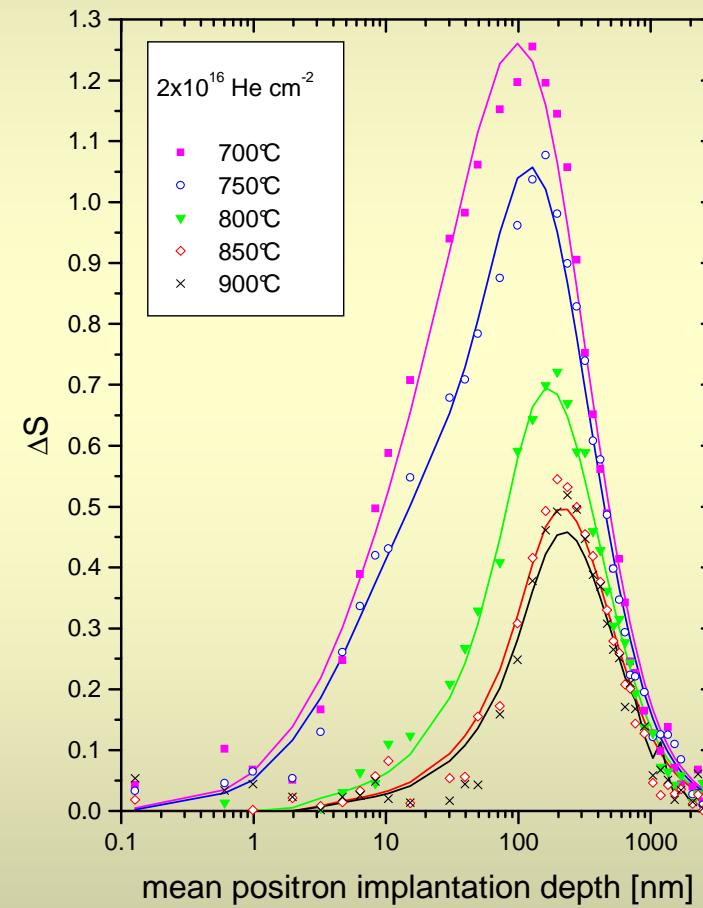
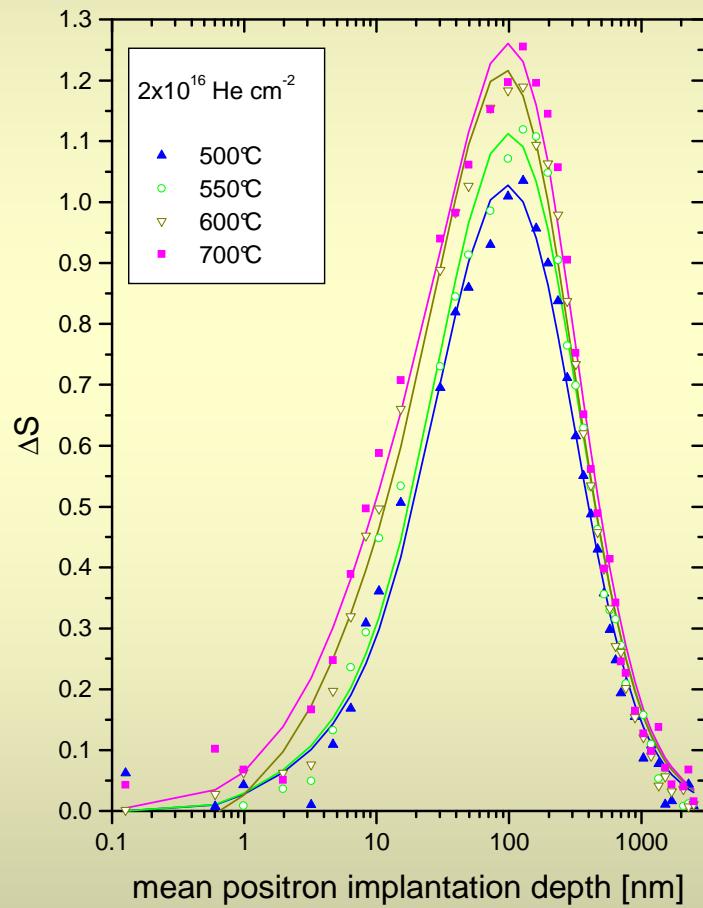
$n=0.5 \times 10^{16}\text{cm}^2$ **NO!**

$n=2 \times 10^{16}\text{cm}^2$ **YES!**

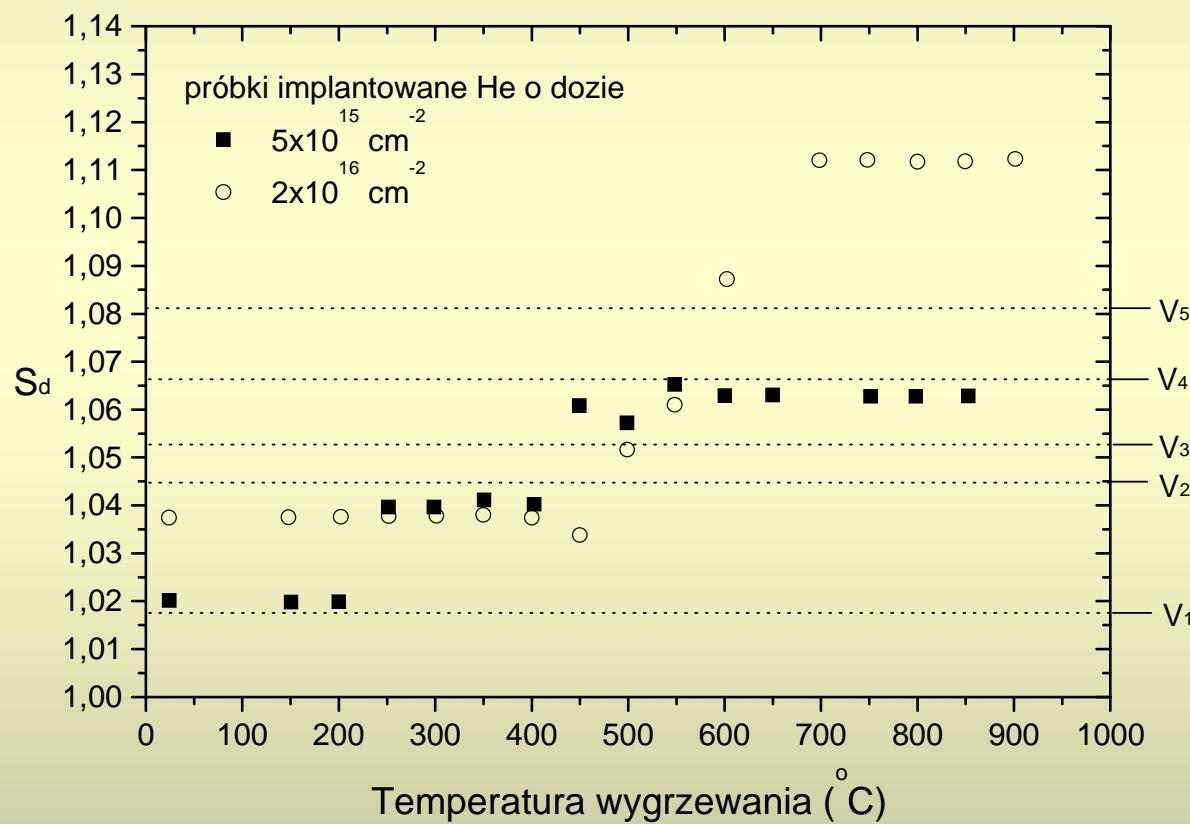
He bubbles in Si



He bubbles in Si

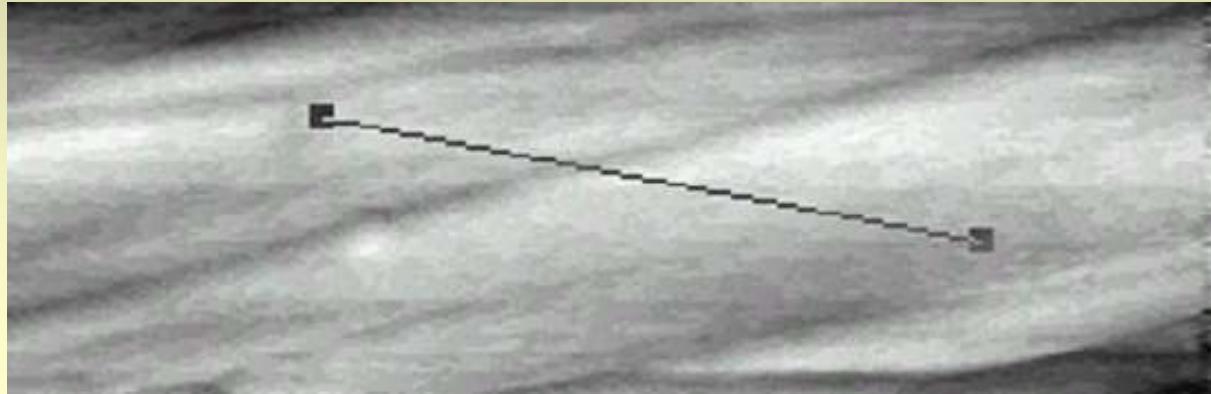


He bubbles in Si quantization of S - values

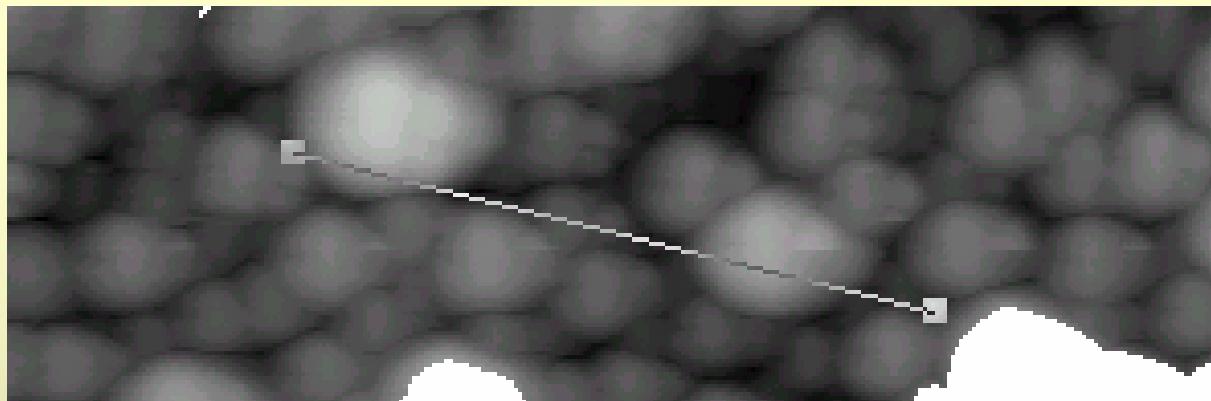


Metal-doped glasses (Pb, Bi)

a)



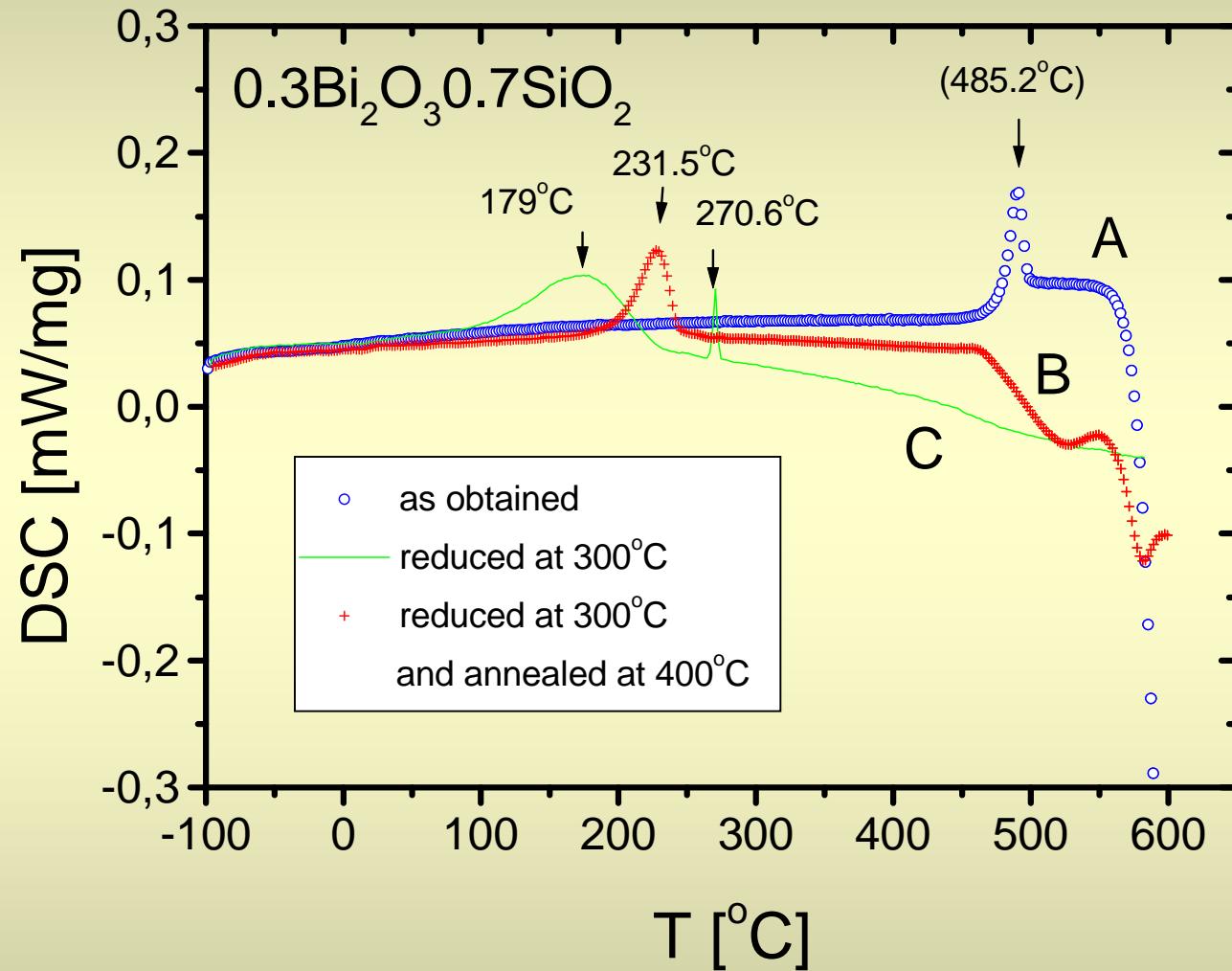
b)



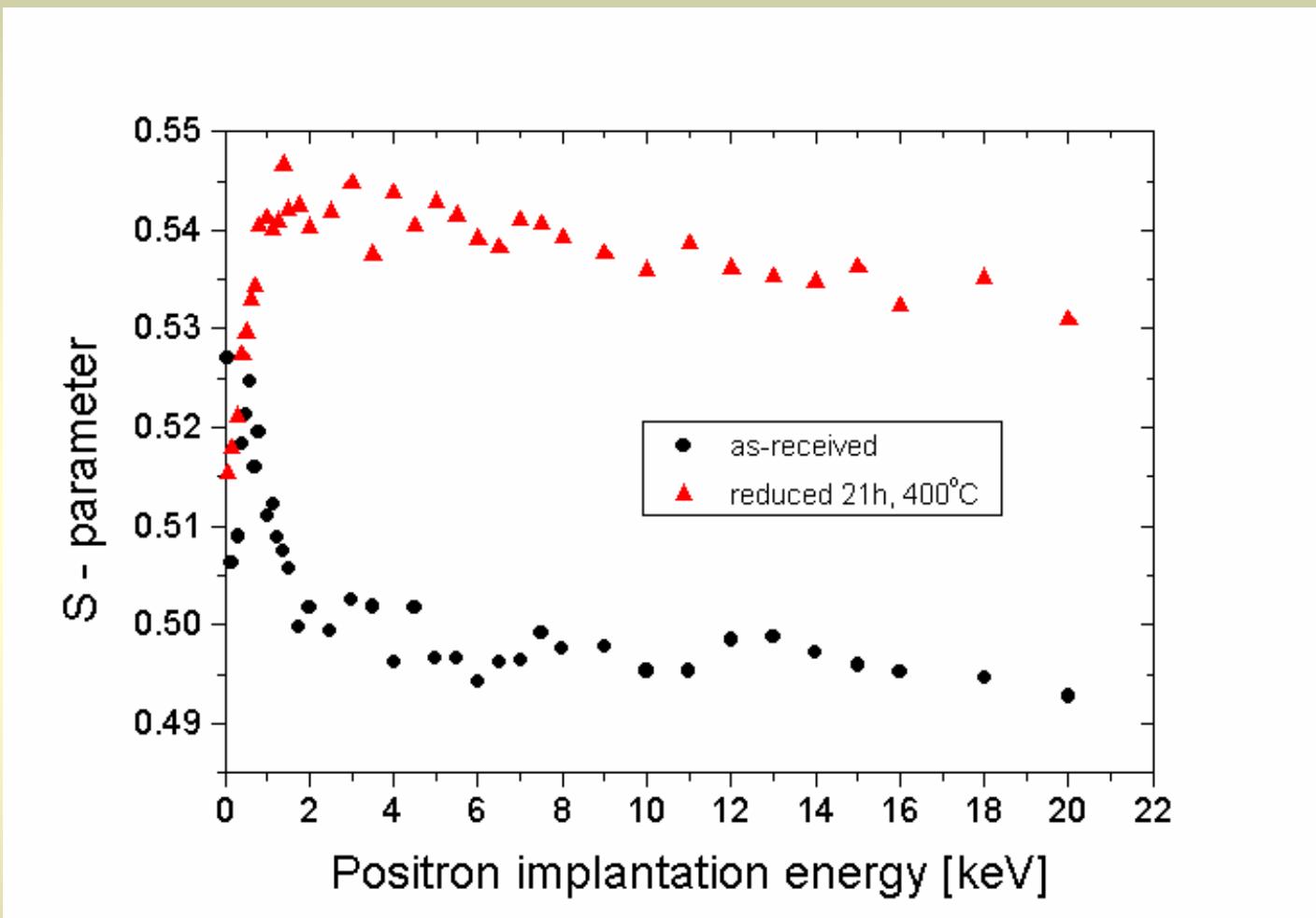
AFM picture of Si-Pb glass; a) freshly broken;

b) Annealed at 580°C for 21h

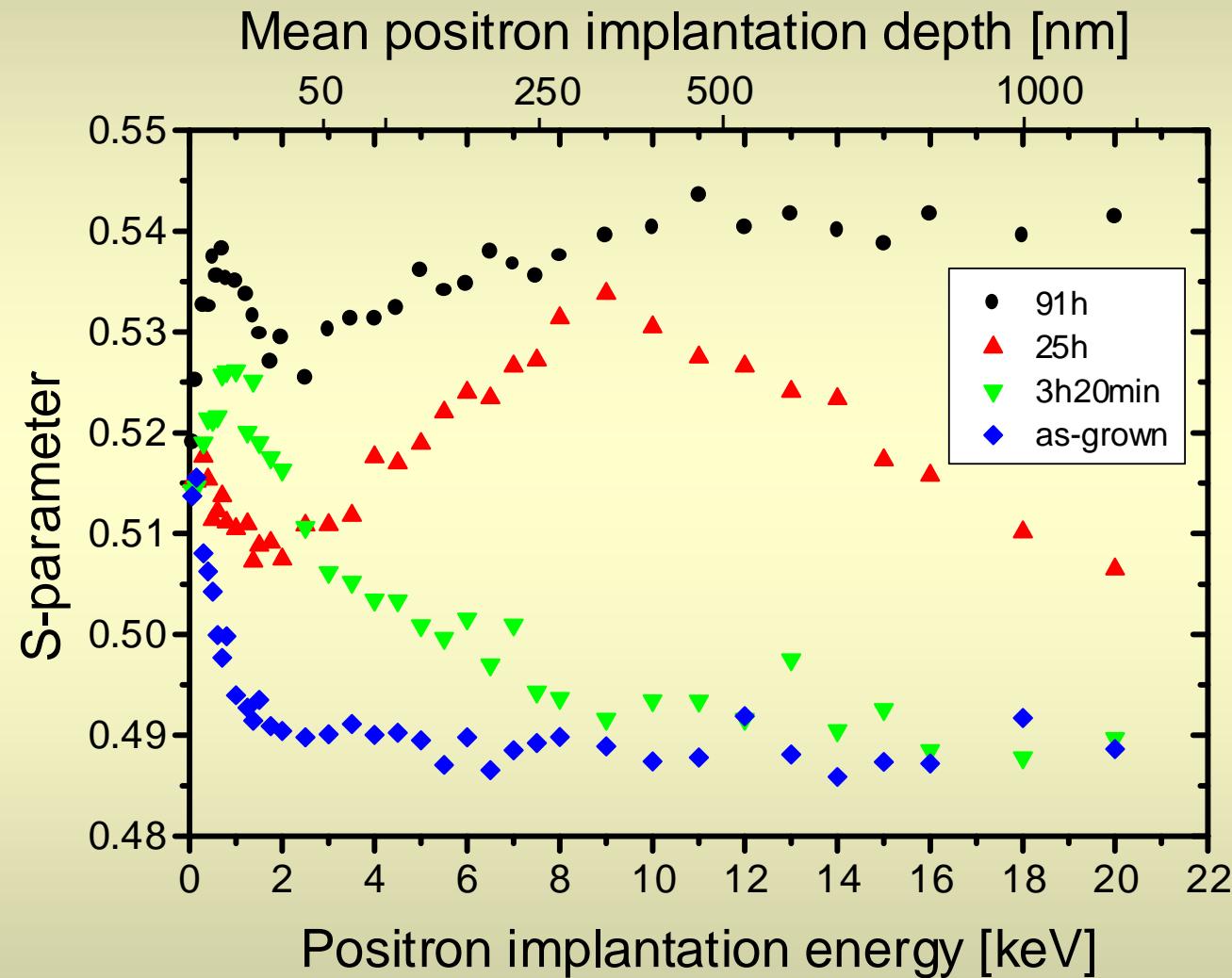
Metal-doped glasses ($\text{SiO}_2 + \text{Bi}_2\text{O}_3$)



Conducting glasses ($\text{SiO}_2 + \text{PbO}_2$)

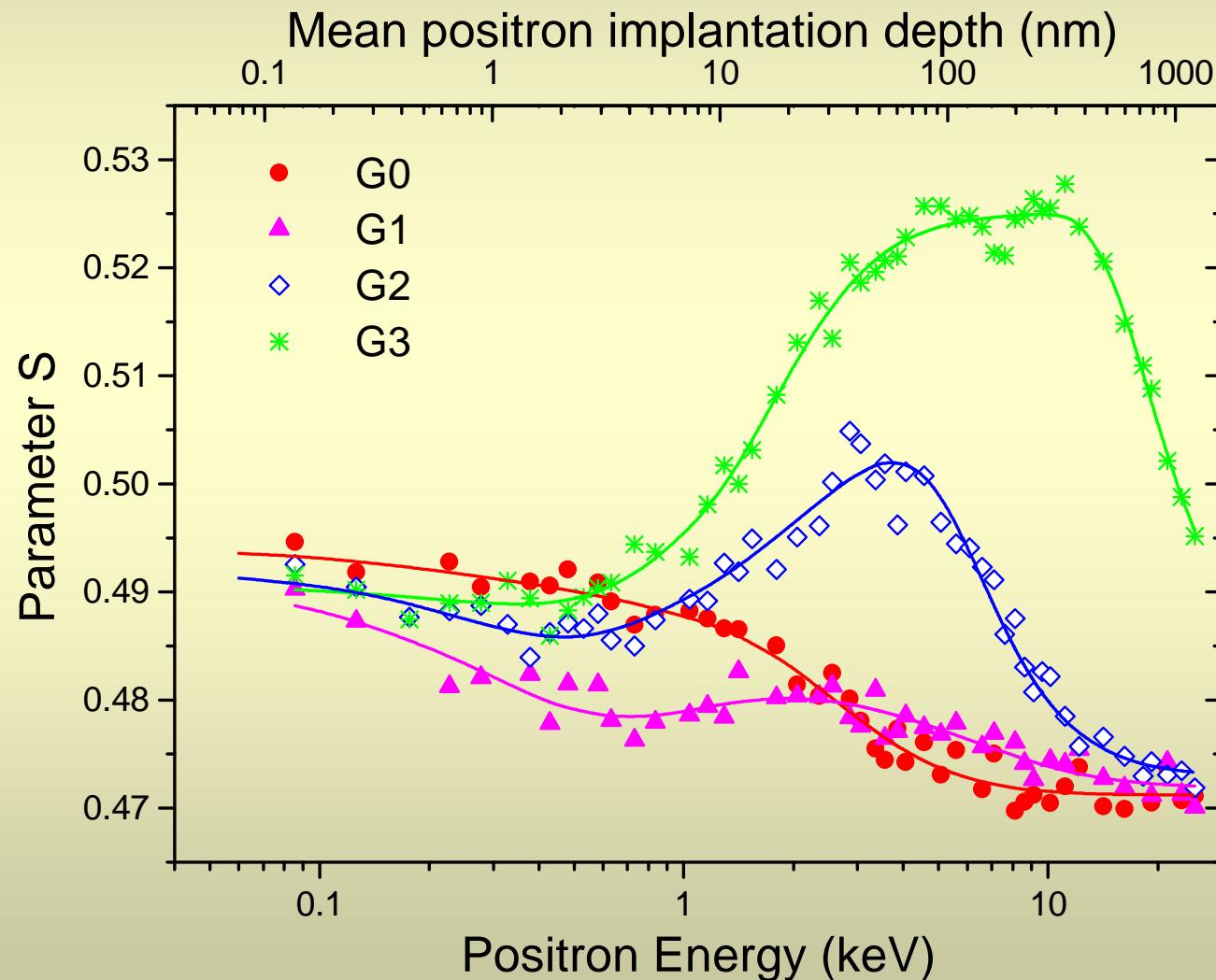


Doped glasses ($\text{SiO}_2 + \text{Bi}_2\text{O}_3$)



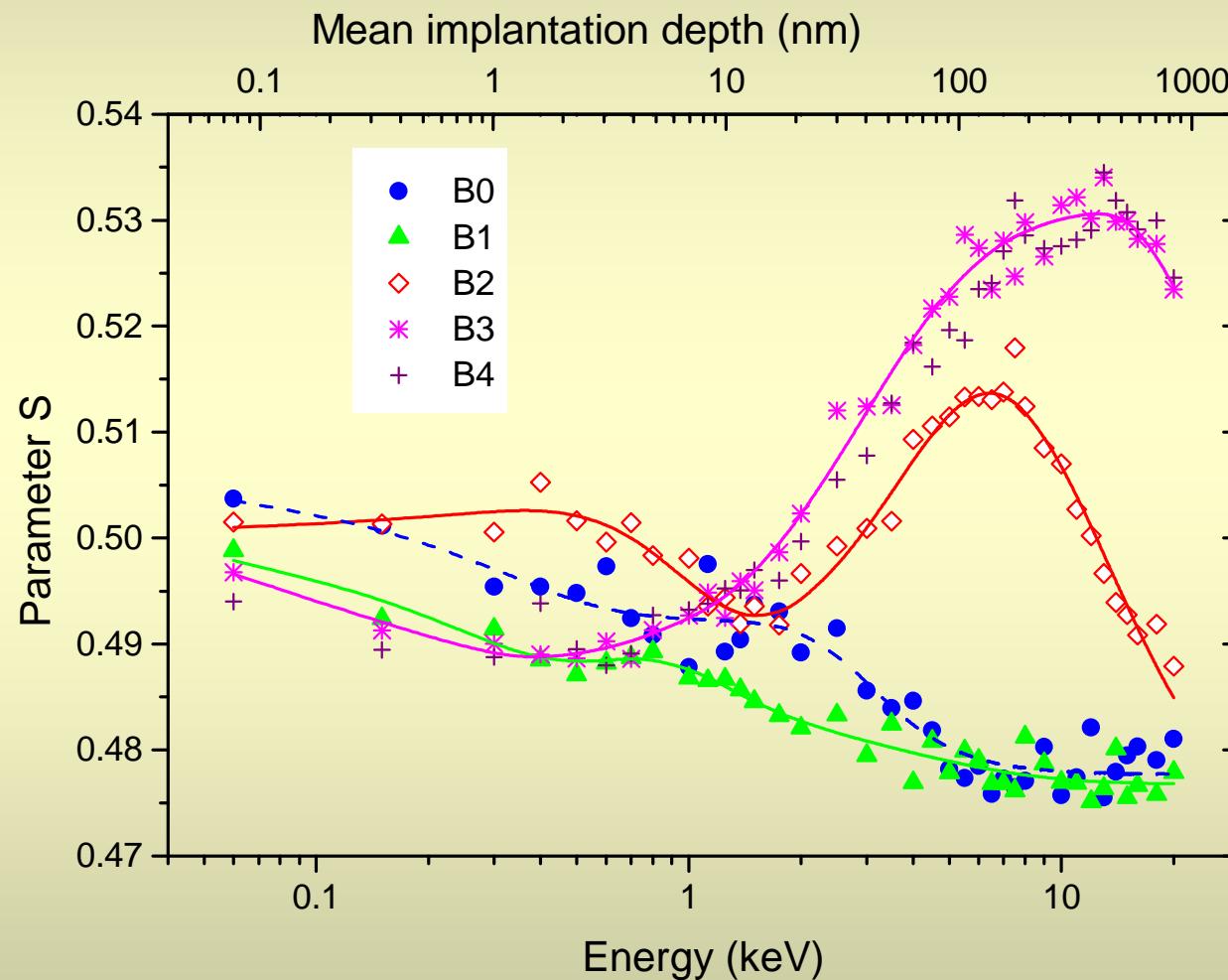
Conducting glasses ($\text{GeO}_2 + \text{Bi}_2\text{O}_3$)

Metal precipitates – depth profiling

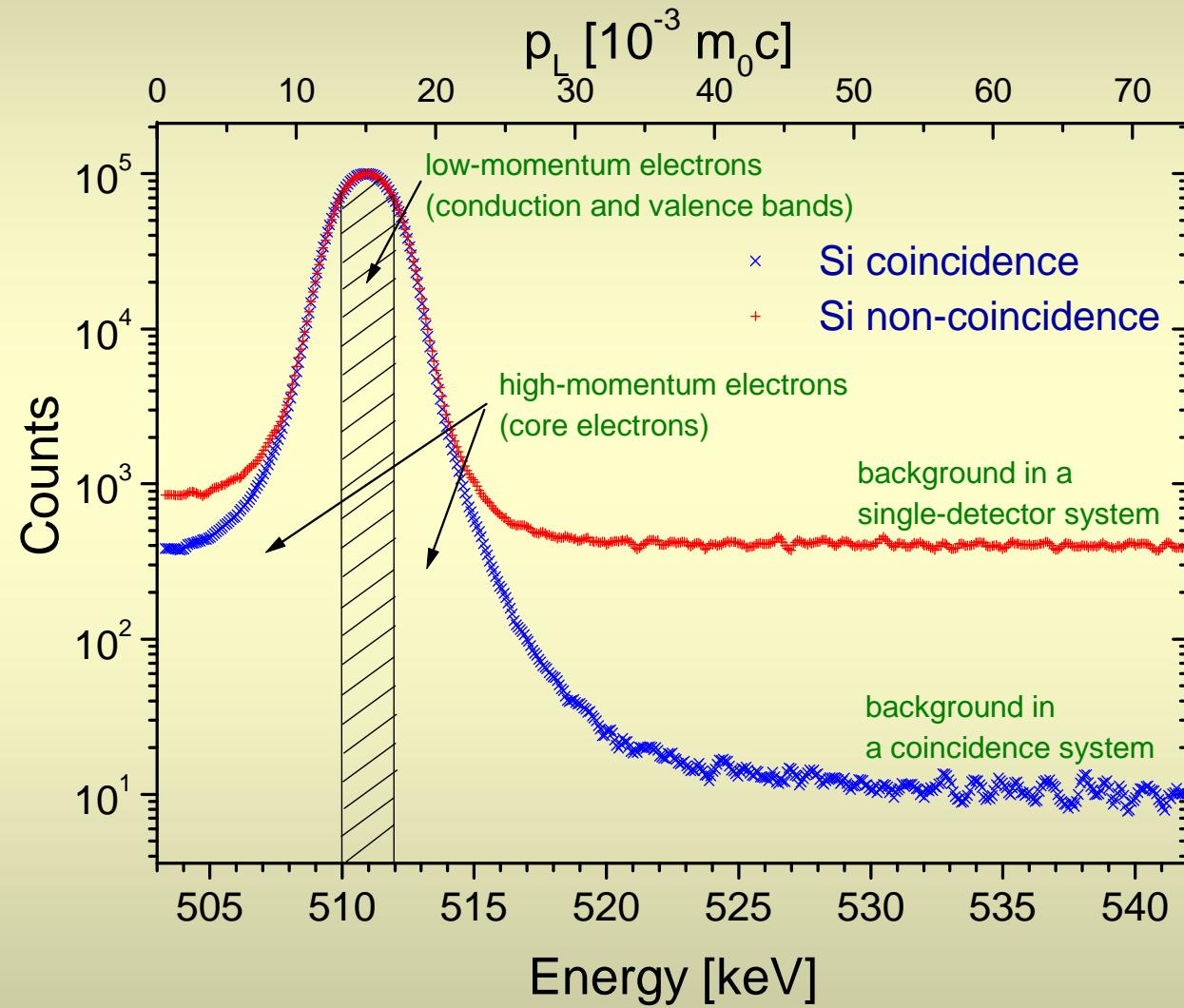


Conducting glasses ($\text{SiO}_2 + \text{Bi}_2\text{O}_3$)

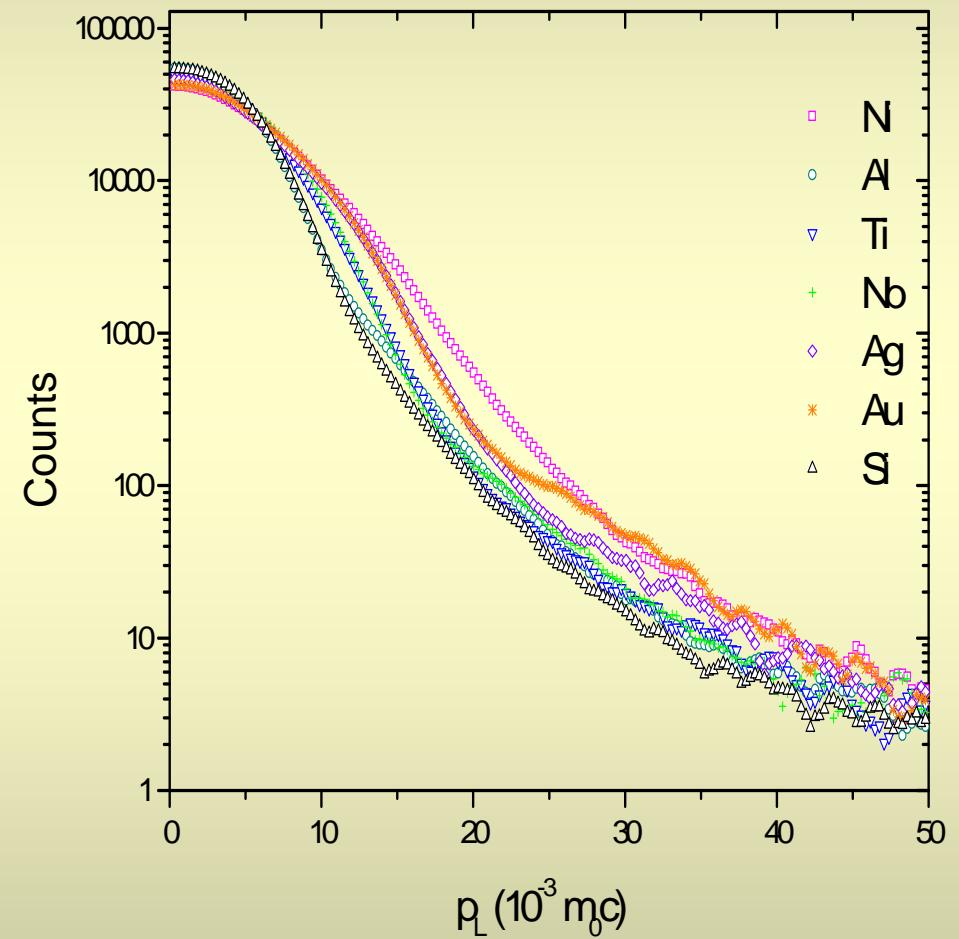
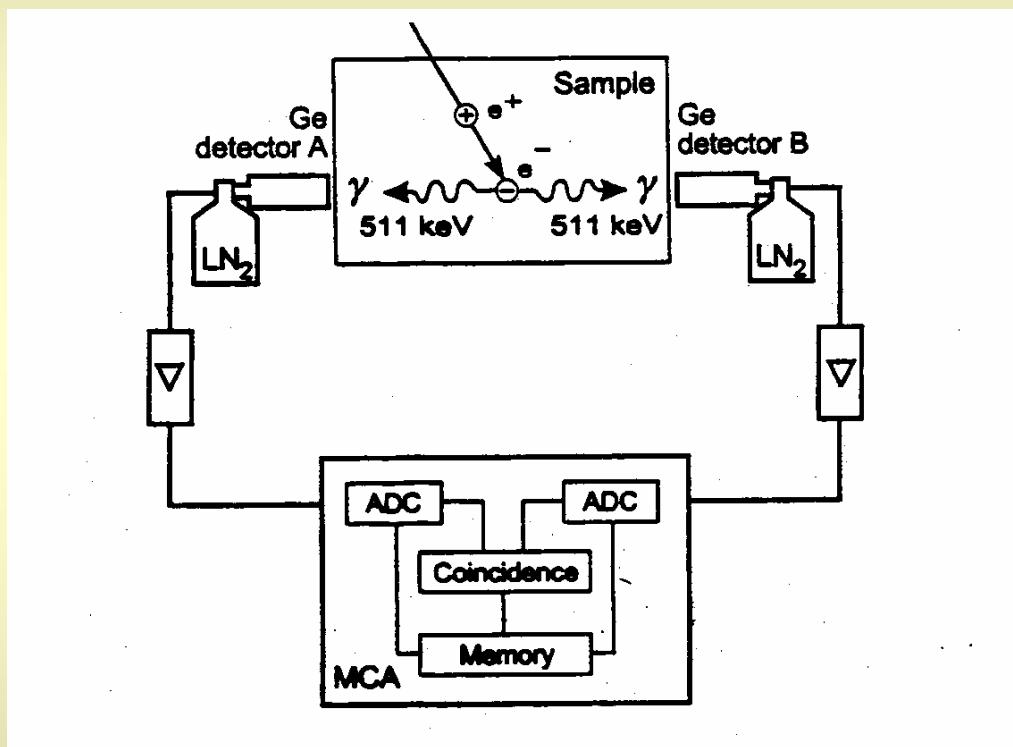
Metal precipitates – depth profiling



3. Doppler-coincidence technique

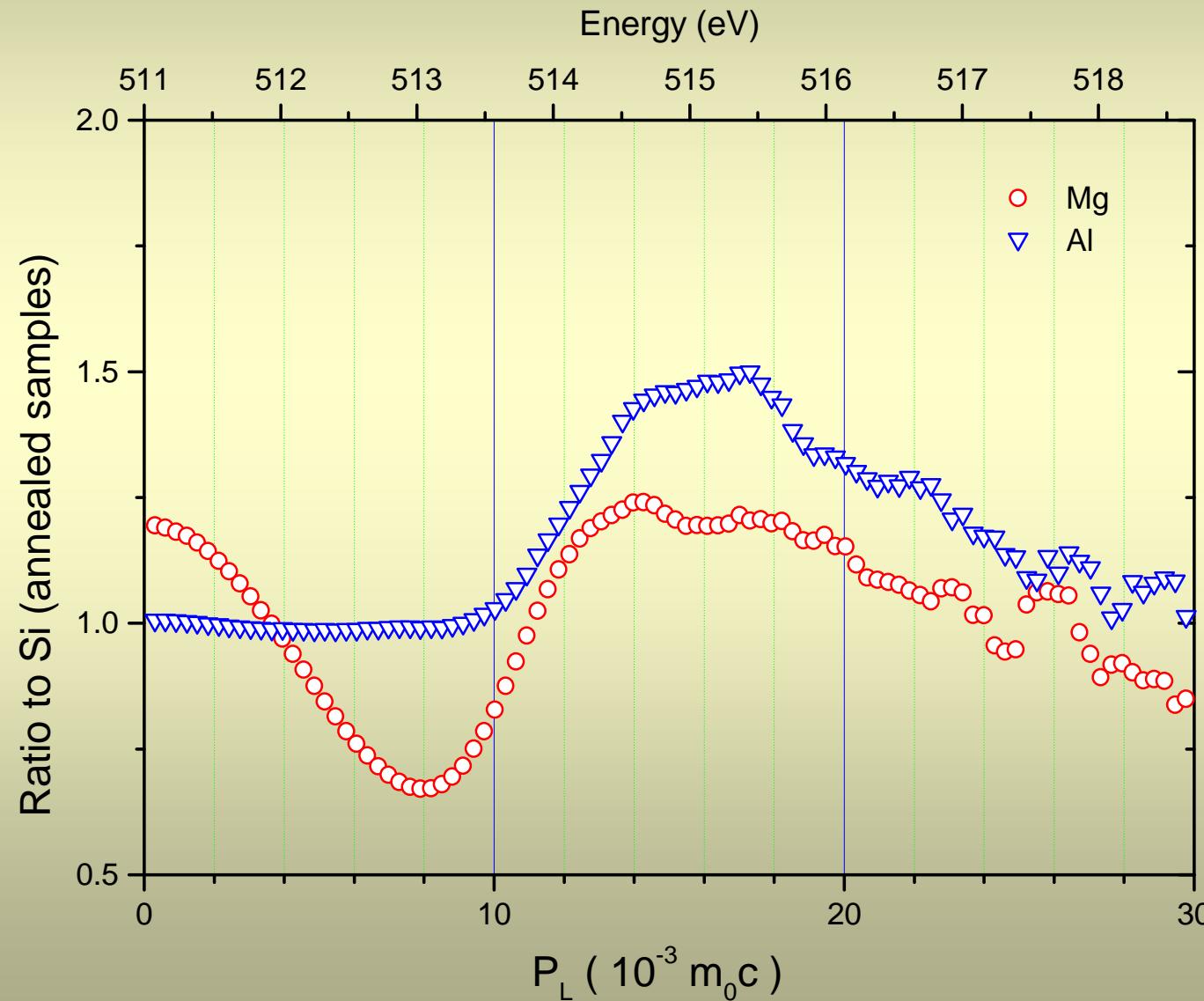


Doppler-coincidence spectra

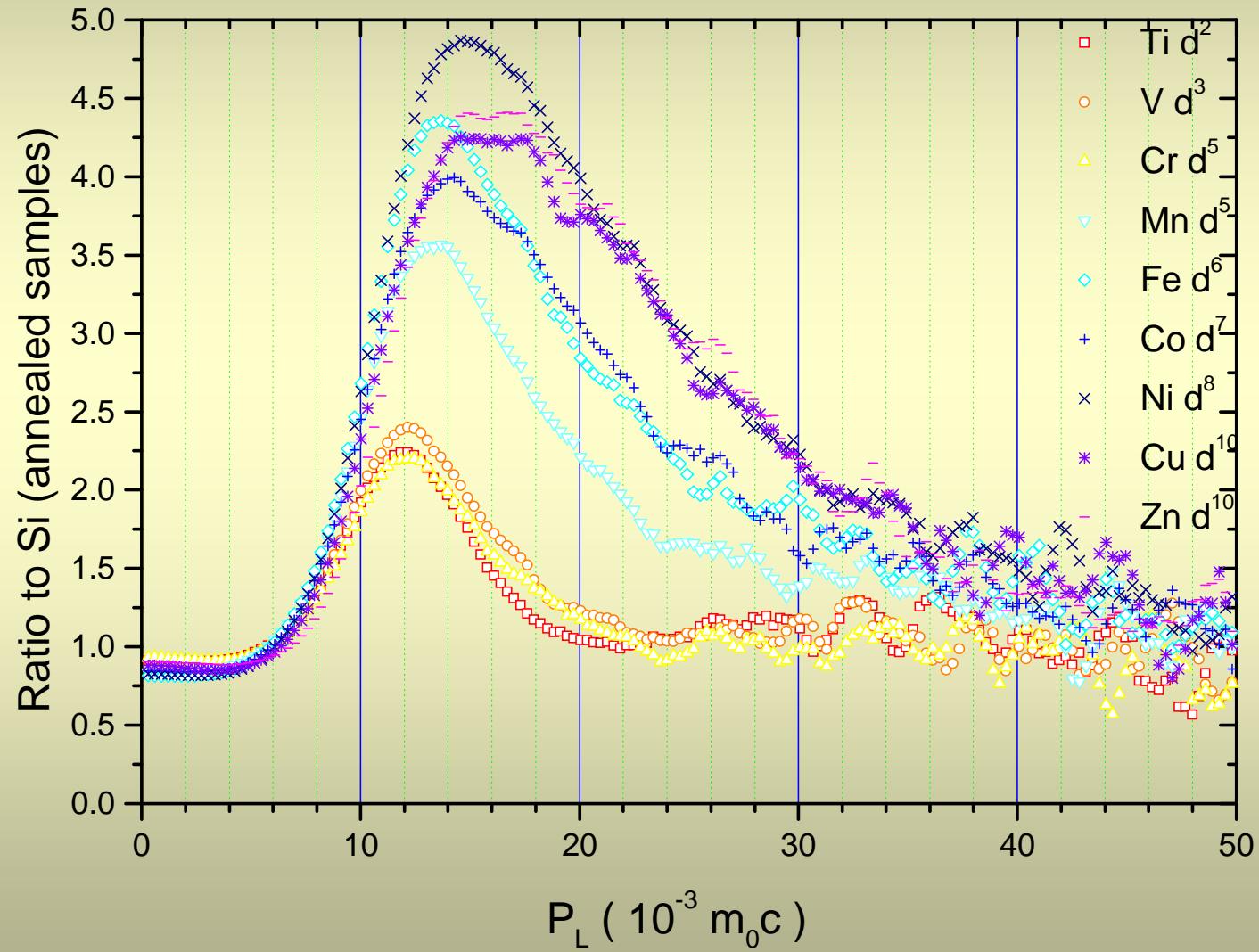


rys

Doppler coincidence – element fingerprints

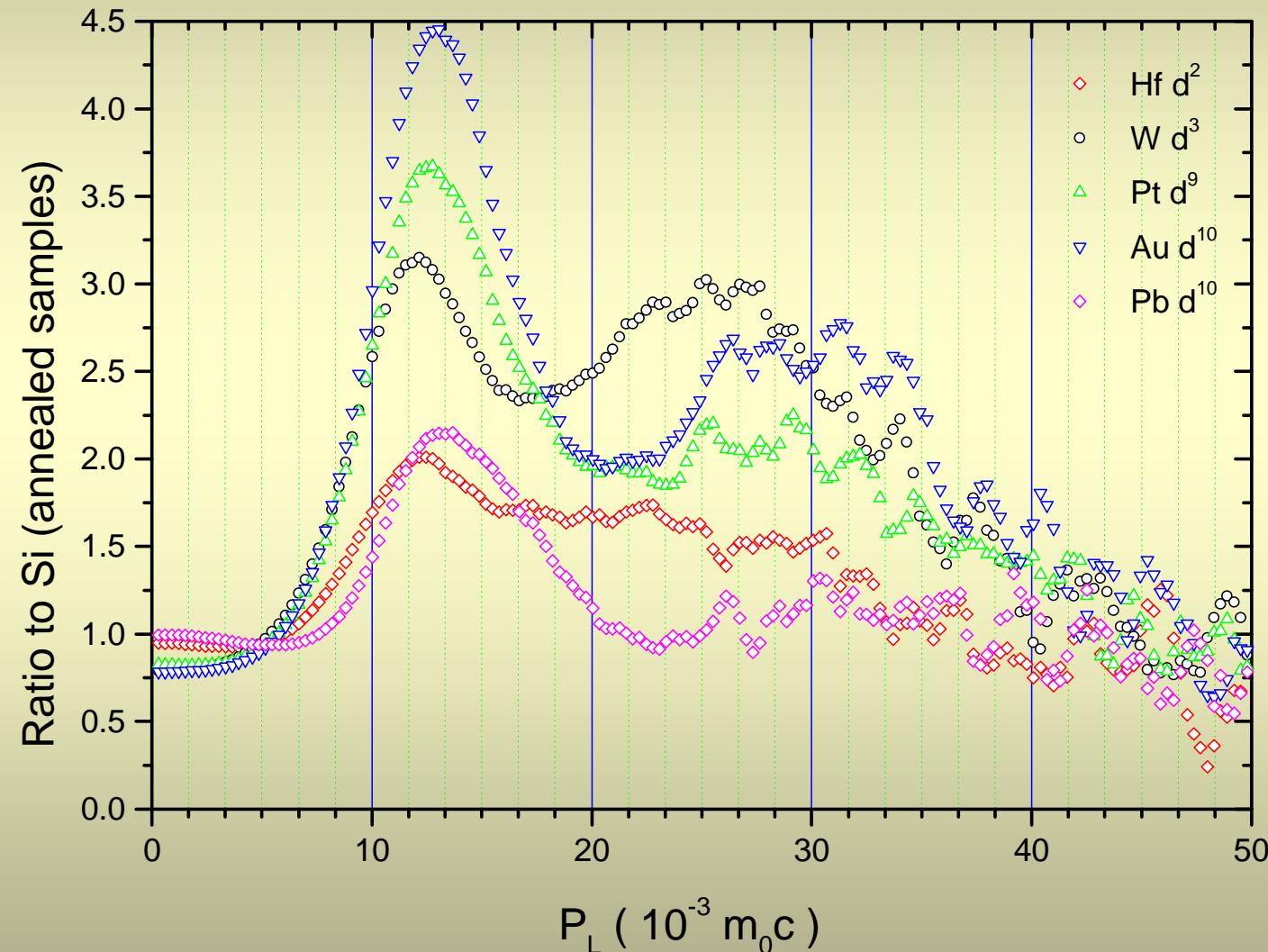


Doppler coincidence – element fingerprints



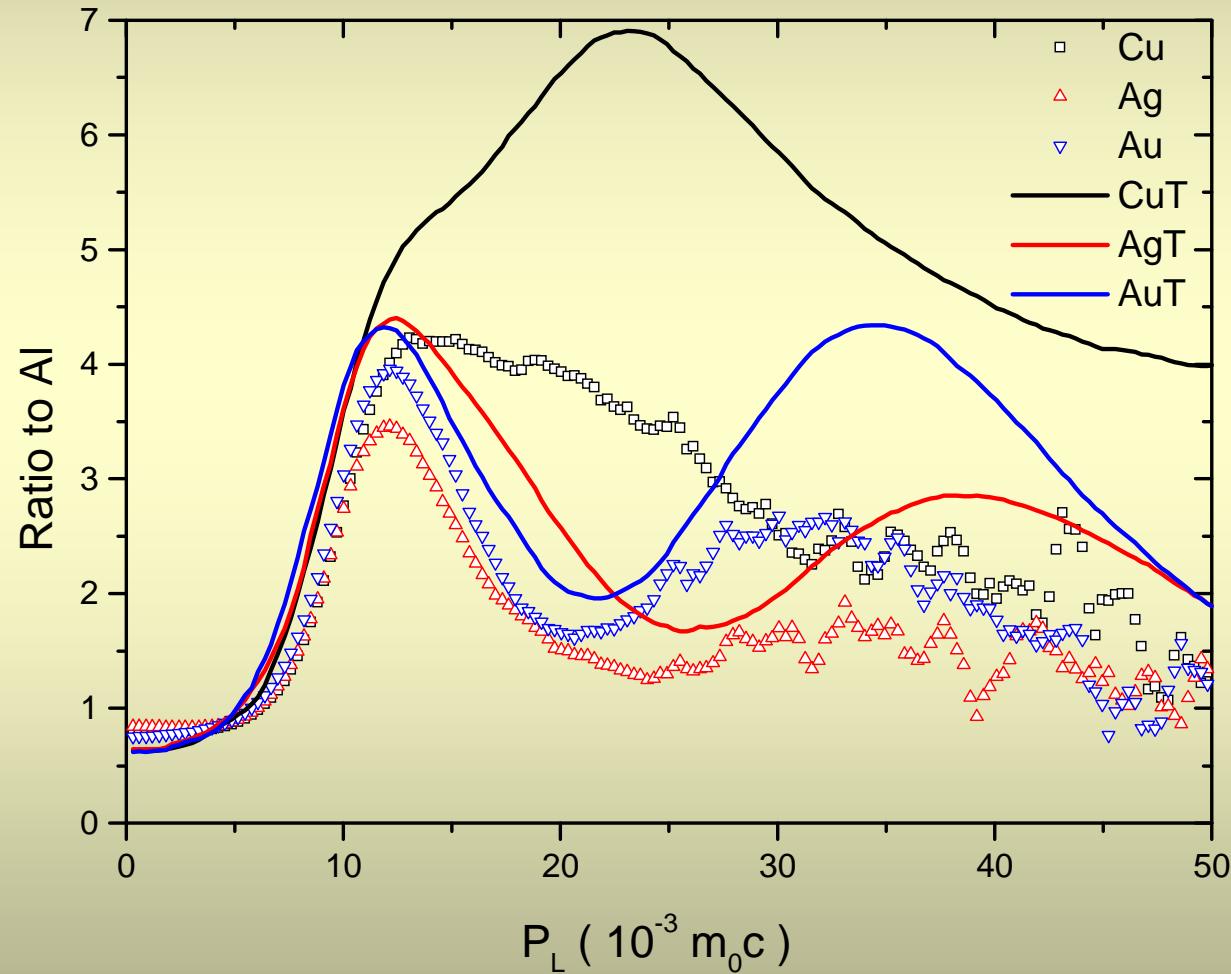
rys5wr2

Doppler coincidence – element fingerprints

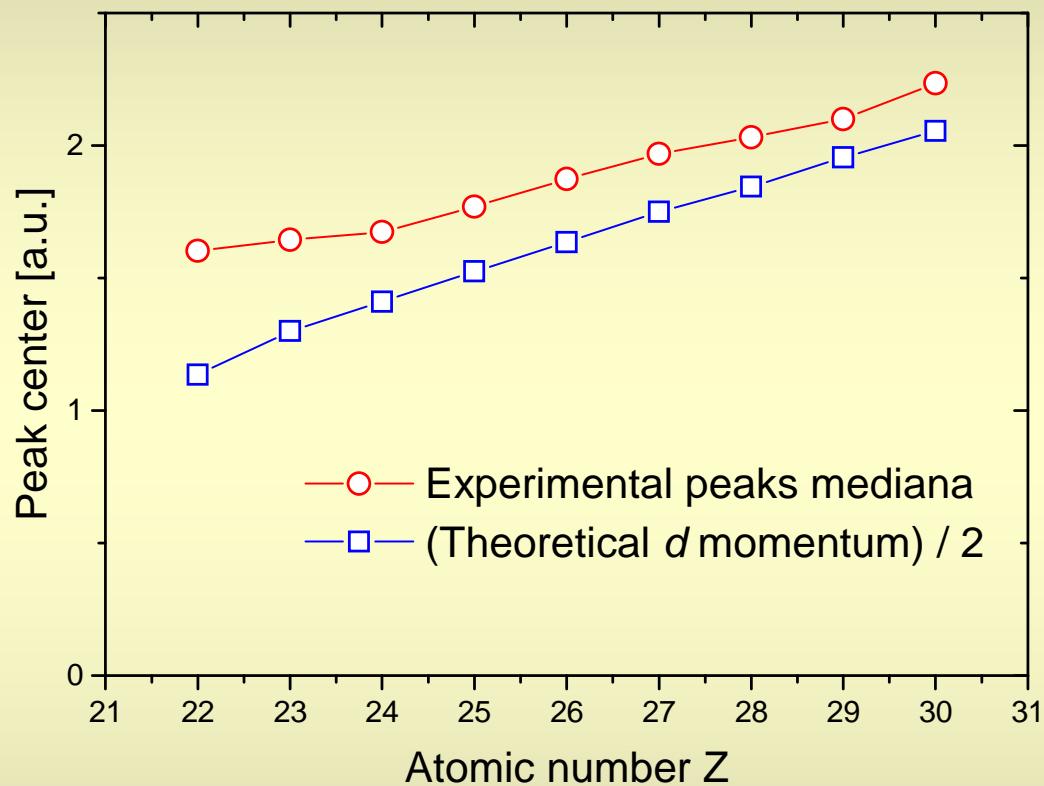


Fingerprints - theory

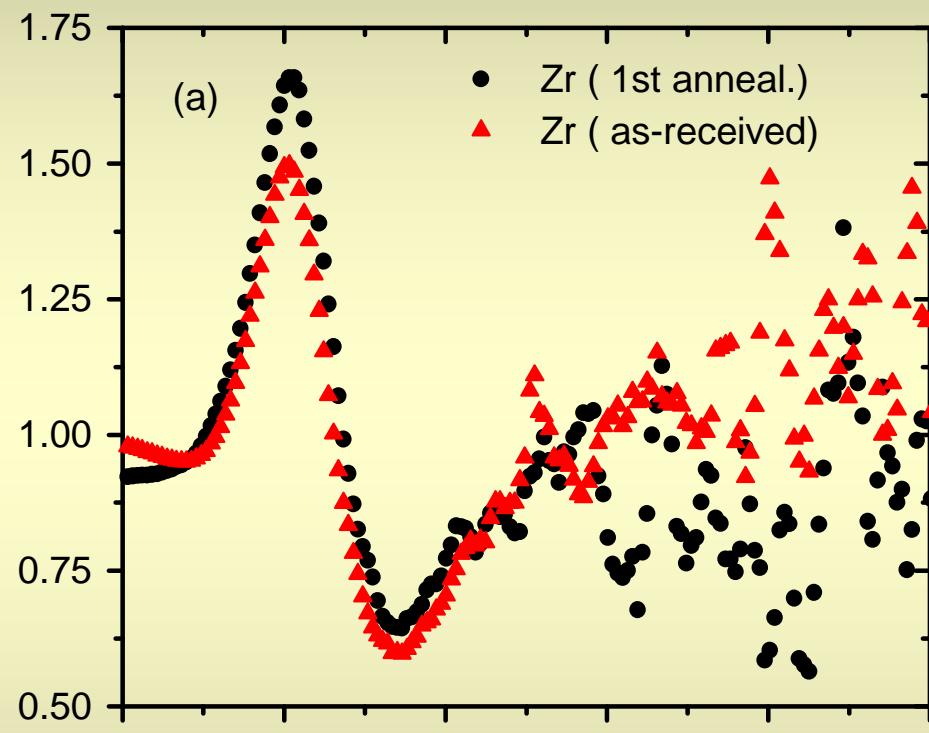
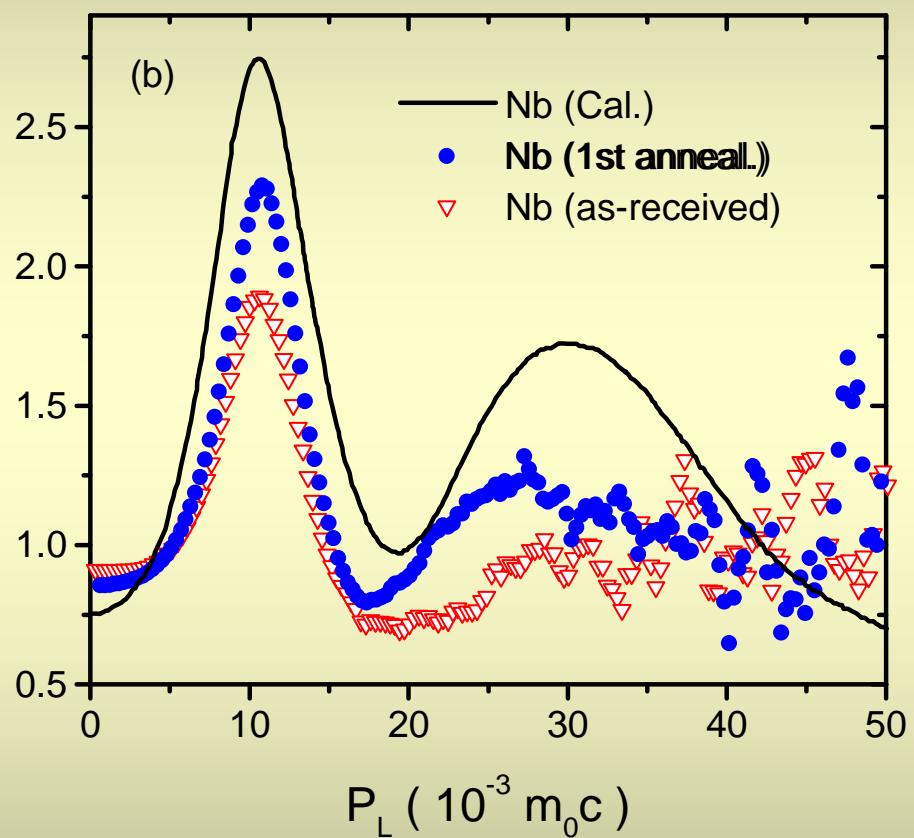
Noble metals



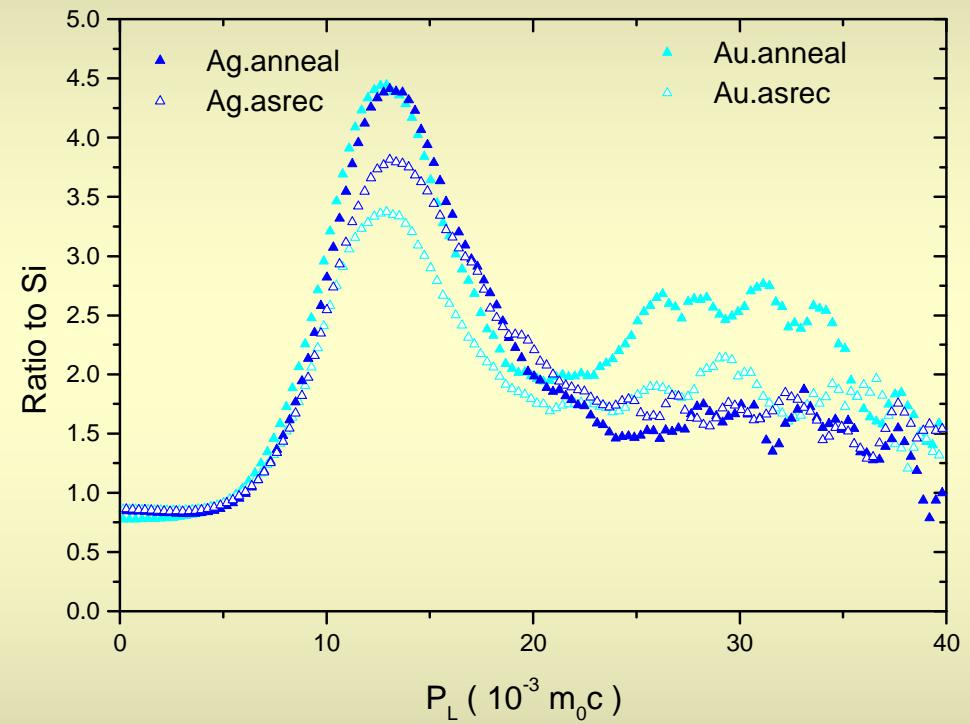
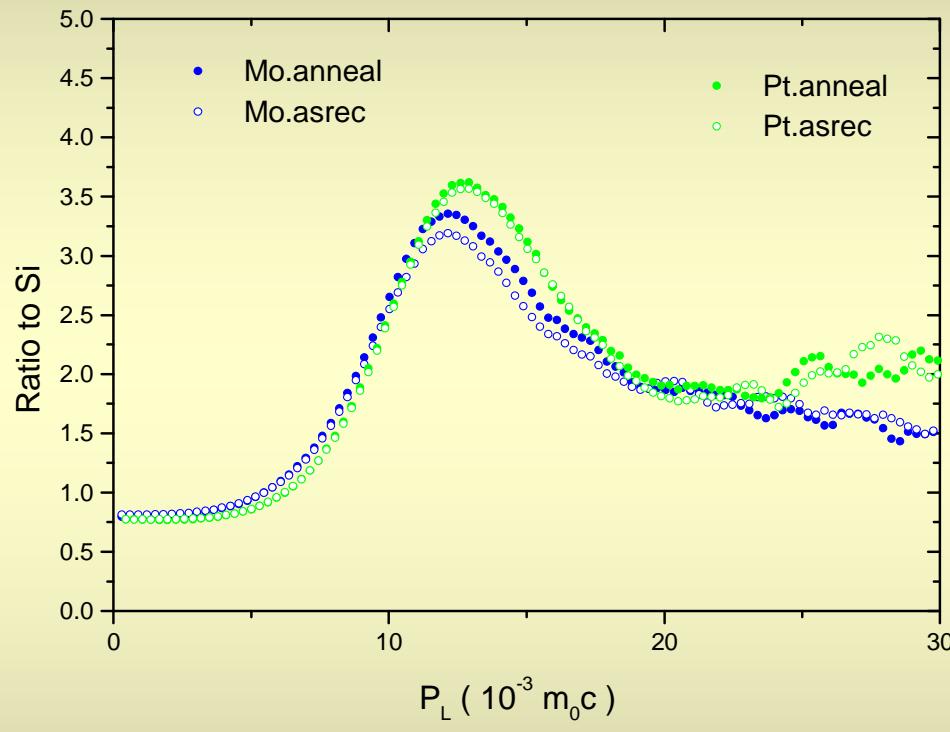
Core-electron momenta



Doppler coincidence – defects



Doppler coincidence – defects

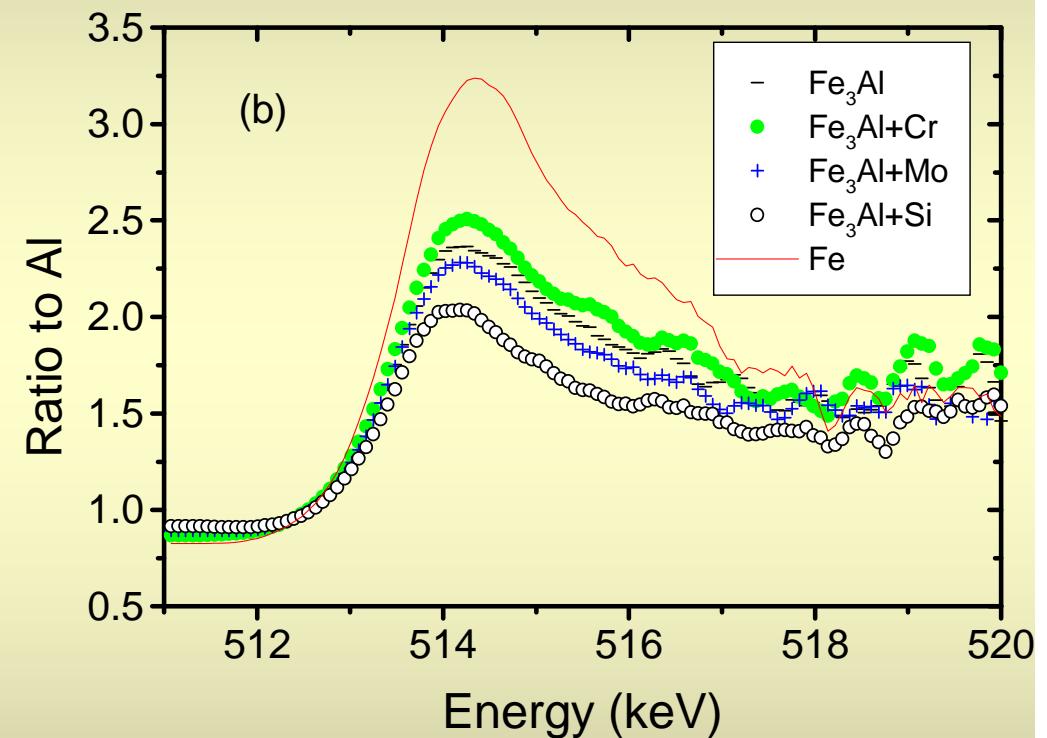
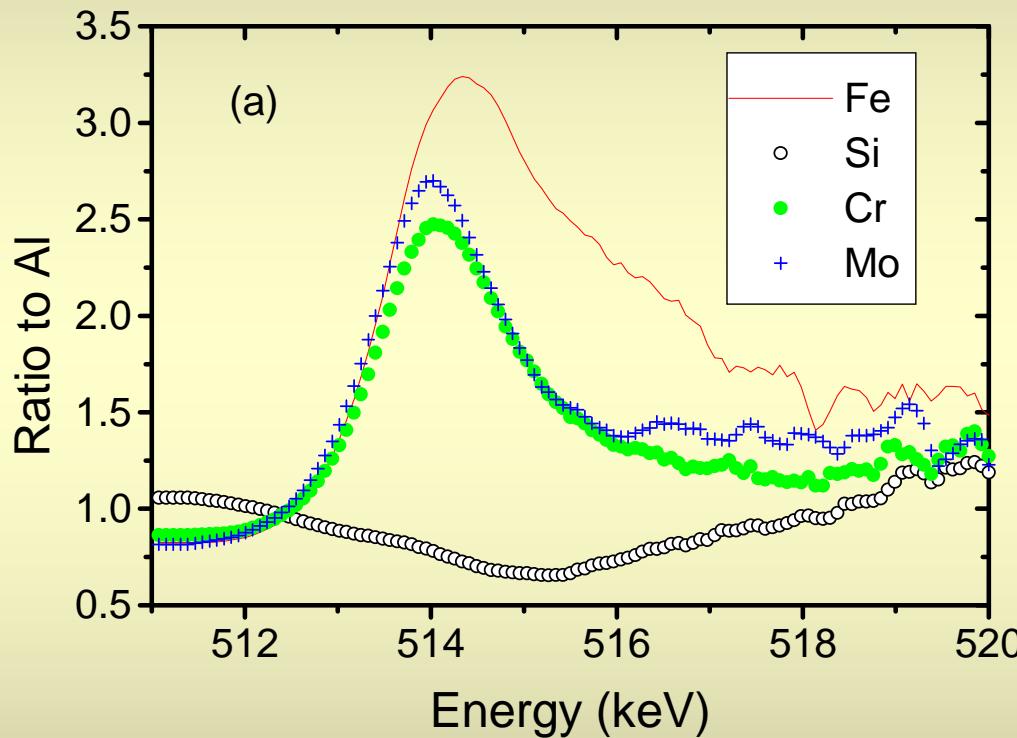


Life-time technique

High-purity metals (annealing)

Z	Elem- ent ^{a)}	Purity	Temp. °C ^{b)}	τ_m [ps]	τ_1 ^{c)} [ps]	τ_2 [ps]	I_2 ^{d)} [%]	Theory [20]
28	Ni	99.98	1000	105	-	-	-	96
29	Cu	99.8	850	115	-	-	-	106
40	Zr	99.8	1100	159	-	-	-	159
42	Mo	99.9	> 1500	119 123	- 117.4	- 660±60	- 1.0	111
47	Ag	99.9	850	135 188	- 122.7	- 240±2	- 57.0	120
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78	Pt	99.5	1000	167	142	282±8	18.0	94
79	Au	99.99	850	123 190	119 181	230±56 290±40	3.7 7.9	107
82	Pb	99.5	250	206 210	- 201.3	- 960±75	- 1.2	187

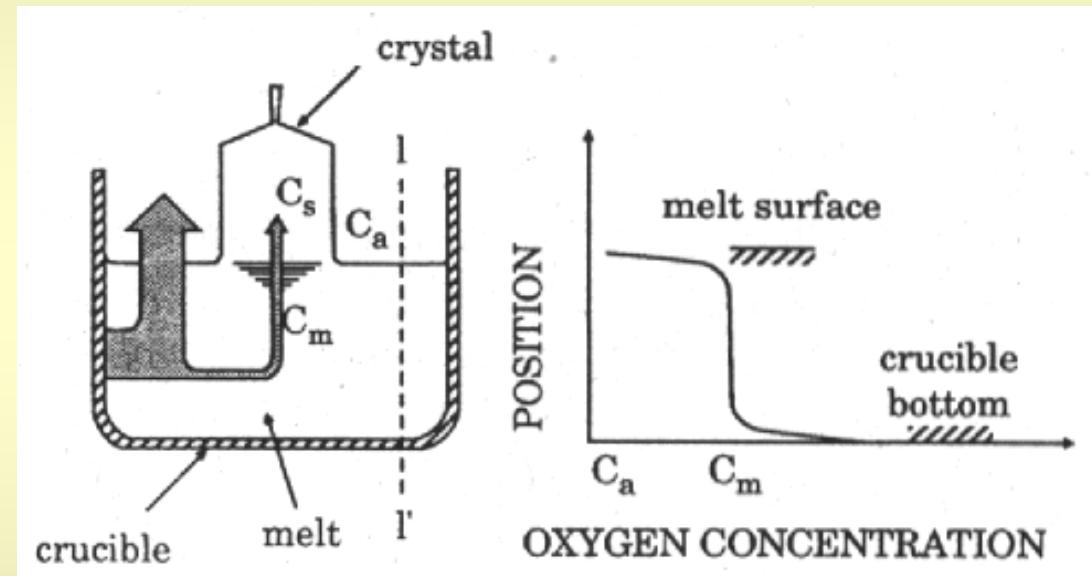
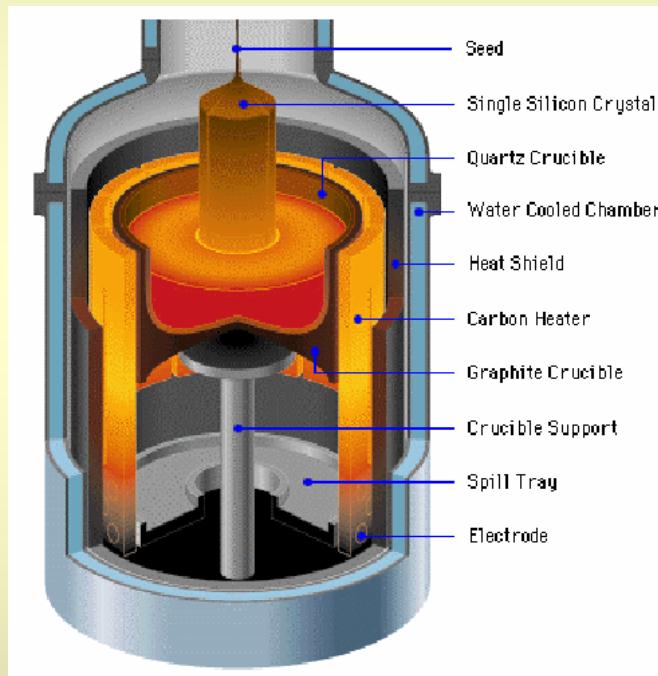
Doppler coincidence – alloys



Not a simple sum!

1+2+3=Combine methods – case study

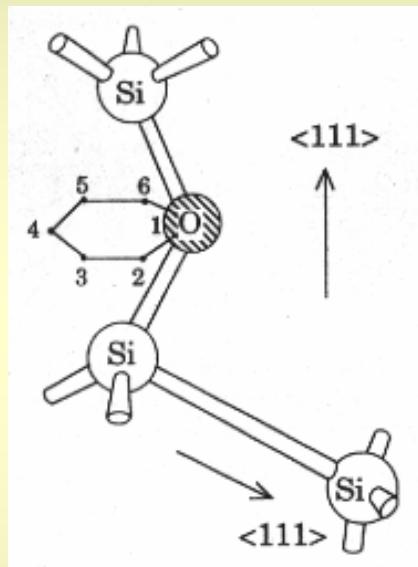
Czochralski-grown Silicon



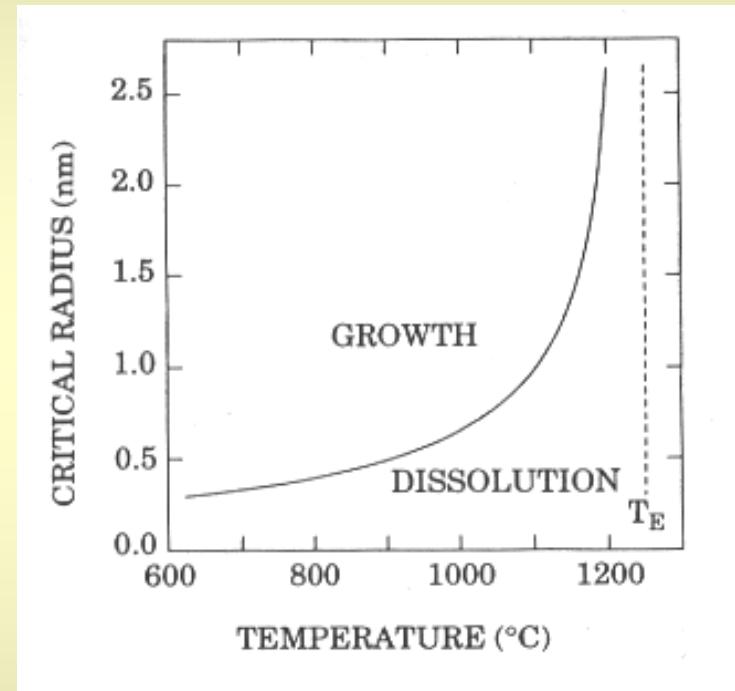
$$c_O \approx 10^{18} \text{ cm}^{-3}$$

$$c_B \approx 10^{16} \text{ cm}^{-3}$$

Oxygen in Cz-grown silicon



thermal donors
precipitates
new donors



“as grown”: annealed at 450°C

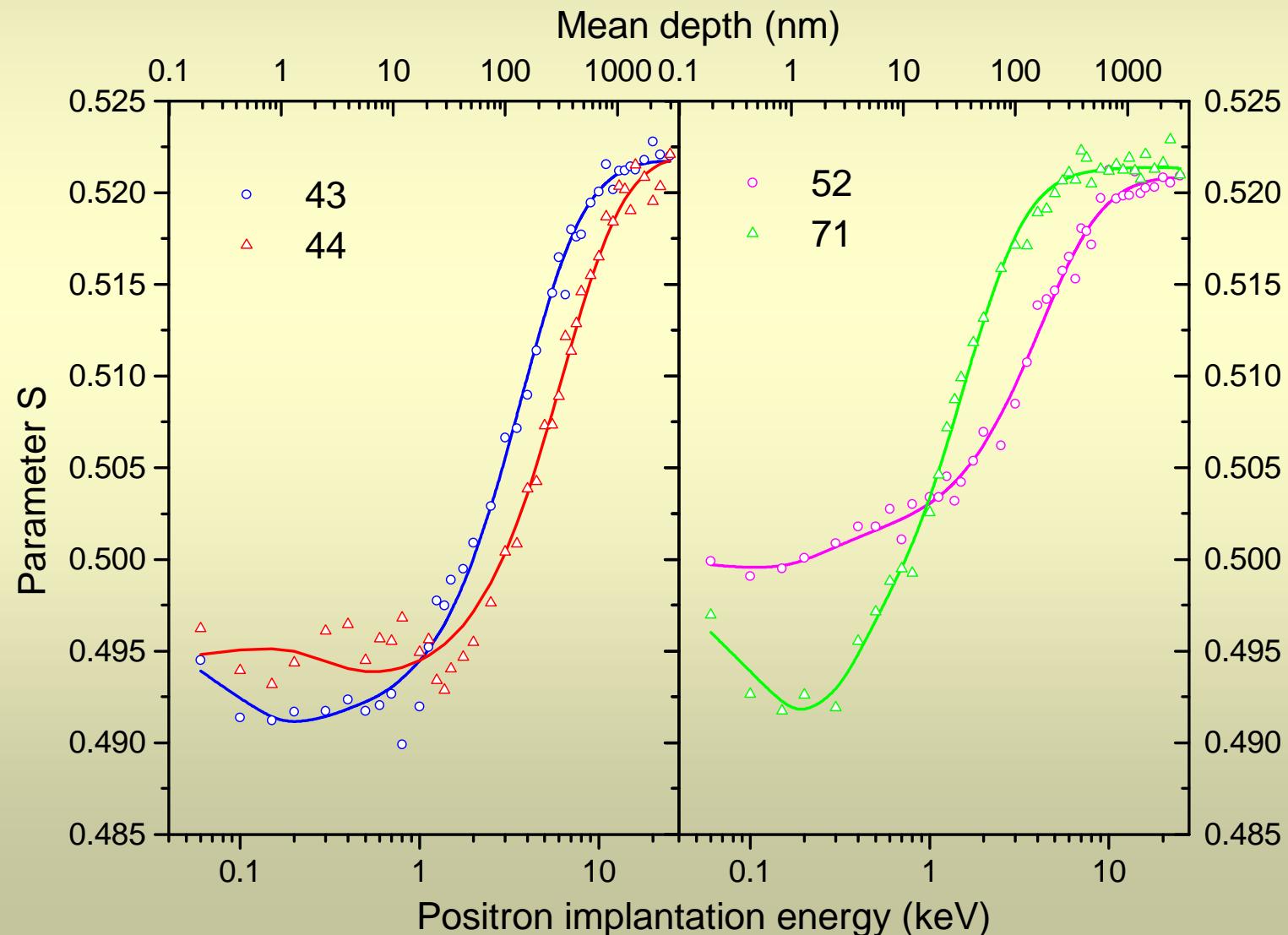
Reference 5Ω 221.6ps±1.0 T_r=1.0ns±0.3 I_r=0.38%±0.12

No	Symbol	Pretreatment		Treatment		eV	PL Int.	Lifetime		Oxygen
		Temp/time	Temp/time	Temp/time/press.	T ₁			T ₂	I ₂	
11	W18	no-pre	-	as grown			222.7	1.3	0.42	
							±0.84	±0.2	±0.11	
51	W2	450/20	-	--			221.8	1.0	0.35	
							±1.1	±0.3	±0.2	
21	W2	450/20	-	450/10h/30b			222.3	1.1	0.38	
							±1.5	±0.2	±0.02	
22	W1	450/10	/	450/10h/11k			222.1	1.3	0.3	
							±1.5	±0.2	±0.04	
62	W18	no-pre	-	450/10h/14kb			223.0	1.0	0.45	
				He			±1.0	±0.1	±0.1	
63	W19	no-pre	-	450/20h/12k			225.0	1.2	0.54	
							±1.2	±0.2	±0.15	
01	W3	450/20	-	600/20h			225.6	1.5	0.33	
ap4	SF3^6	SPD2^2	PDC2^2				±1.2	±0.3	±0.07	3^6
23	W1	450/ 10	-	600/10h/30b			222.2	1.5	0.25	
							±0.3	±0.4	±0.04	
66	W1	450/10		600/10h/6kb			223.8	1.2	0.44	
							±0.8	±2	±0.04	
24	W1	450/10	-	600/10h/10kb			222.5	1.3	0.3	
							±0.4	±0.2	±0.1	
26	W2	450/20		727/5h			not m.			
29	W2	450/20		727/5h/10kb			226.1	1.3	0.52	
							±1.6	±0.2	±0.05	
28	W1	450/10		727/5h/10kb			225.4	1.3	0.61	
							±0.7	±0.2	±0.13	
53	W1	450/10		957/5h			223.8	1.5	0.3	
							±0.8	±2	±0.03	
56	W4	450/20h		1050/20h			220.4	1.1	0.26	6.0
							±0.7	±0.2	±0.1	
03	W4	450/20		1050/20h			221.9	1.8	0.3	45%
ap3	SF6^3	SPD3^2	PDC2^2	Sm5^6			±1.0	±0.5	±0.03	5^6
57	W2	450/20		1127/5h			225.8	1.6	0.49	
							±0.9	±0.4	±0.05	
52	W11	650/20	--				222.6	1.4	0.3	
							±0.5	±0.4	±0.1	
64	W18	no-pre		600/10h/14kb			223.2	1.2	0.3	
							±0.5	±0.3	±0.1	

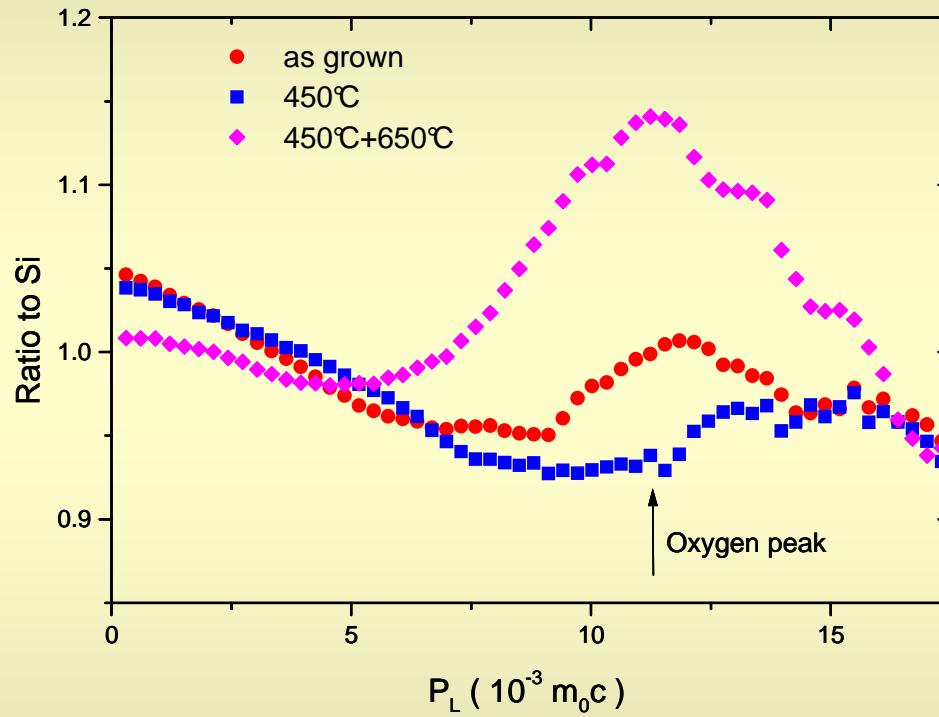
1. Positron lifetime

2. Doppler broadening – positron beam

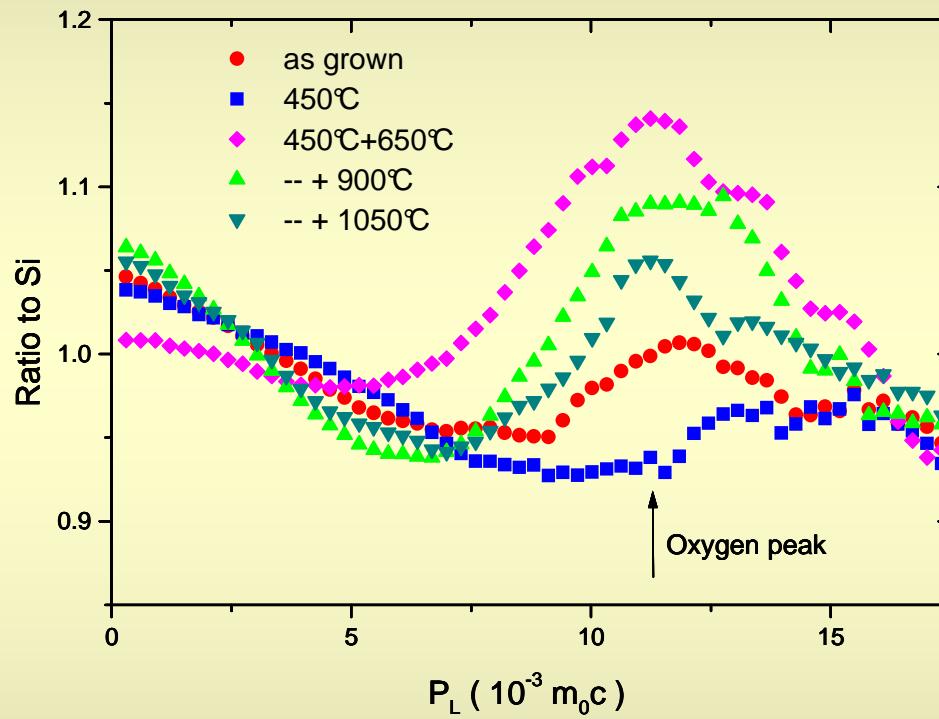
Oxygen in Cz-grown silicon



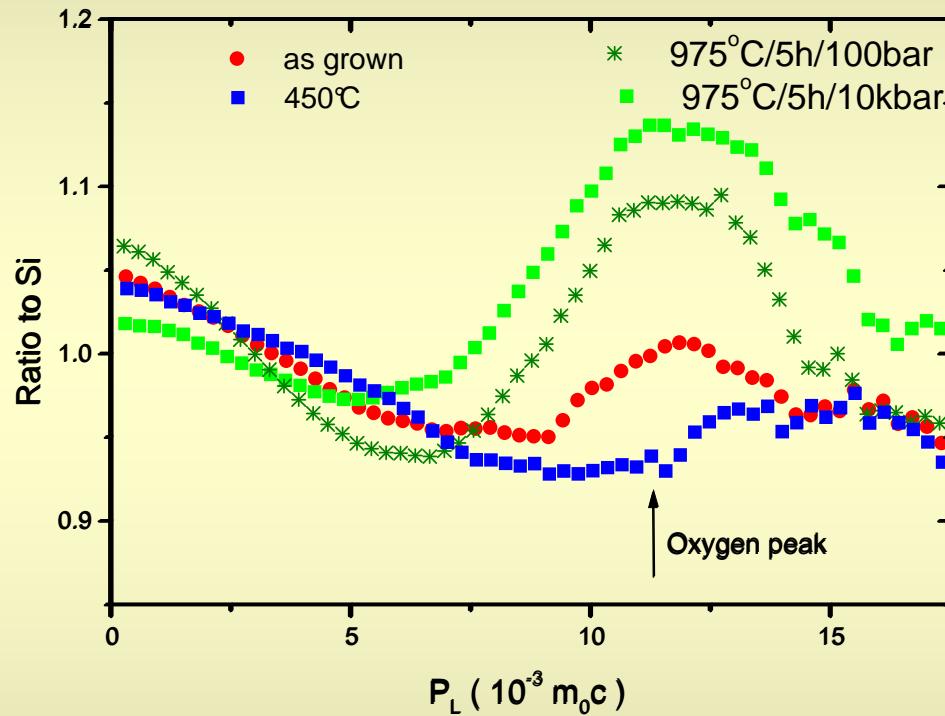
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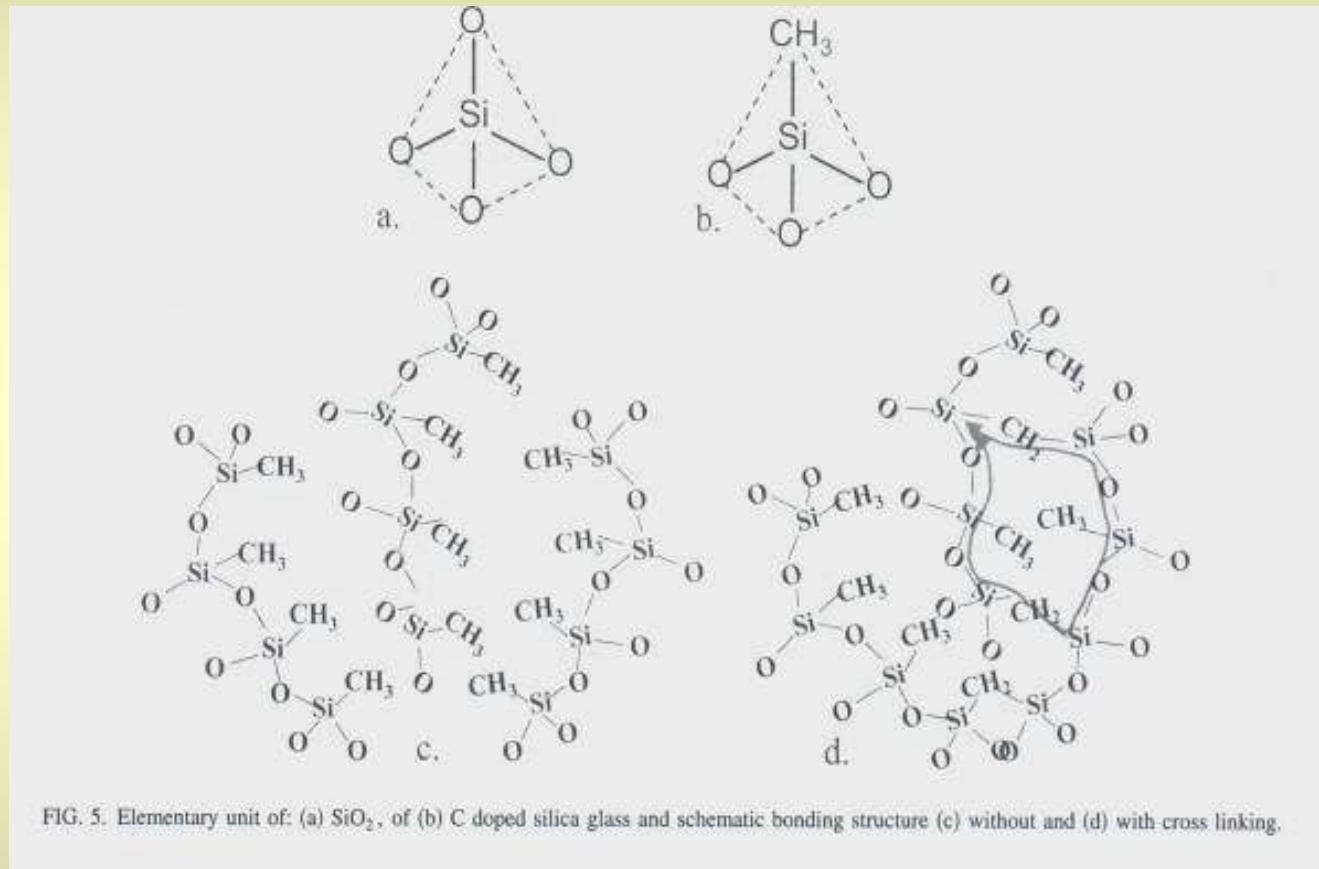
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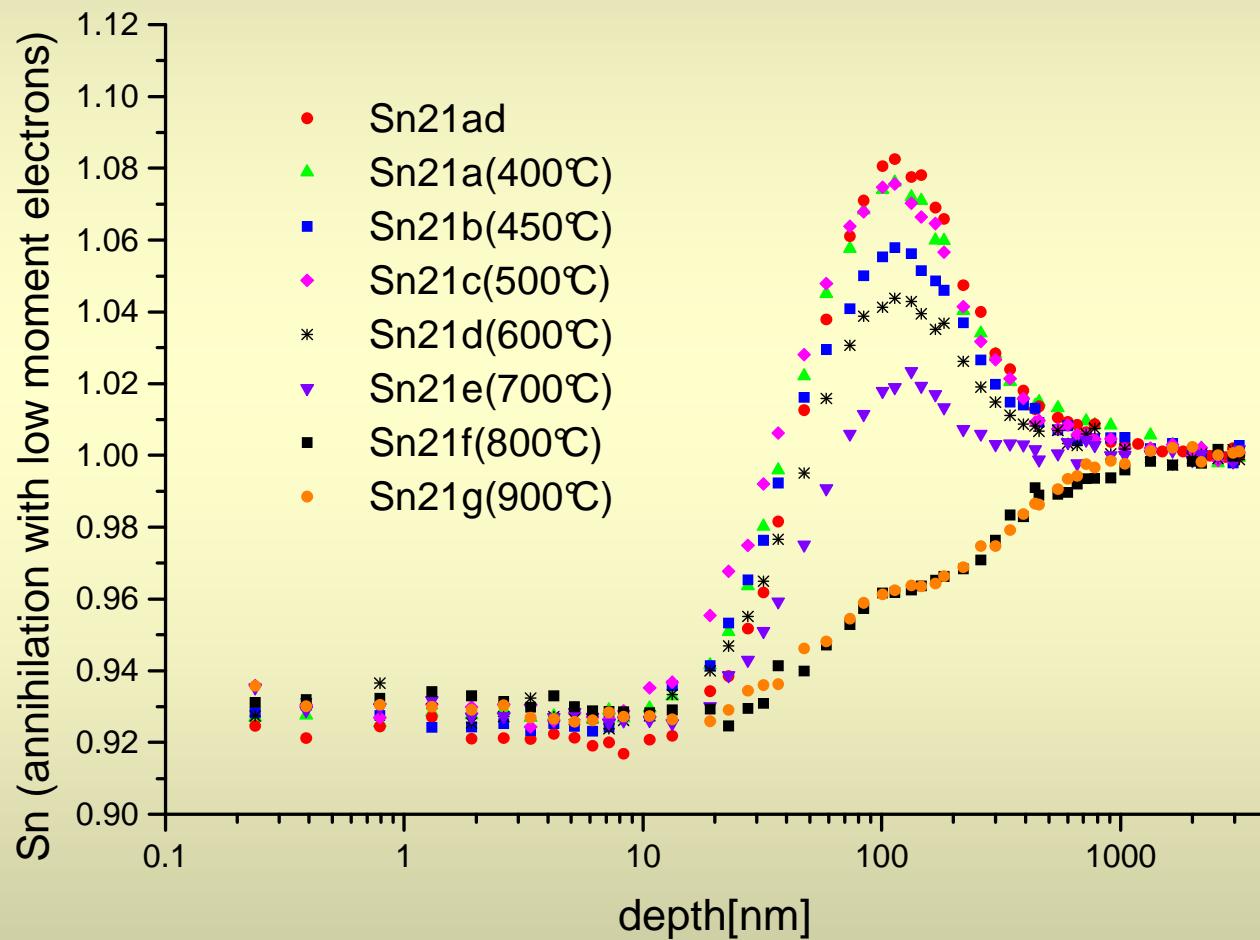


Silica based, low ϵ materials - structure

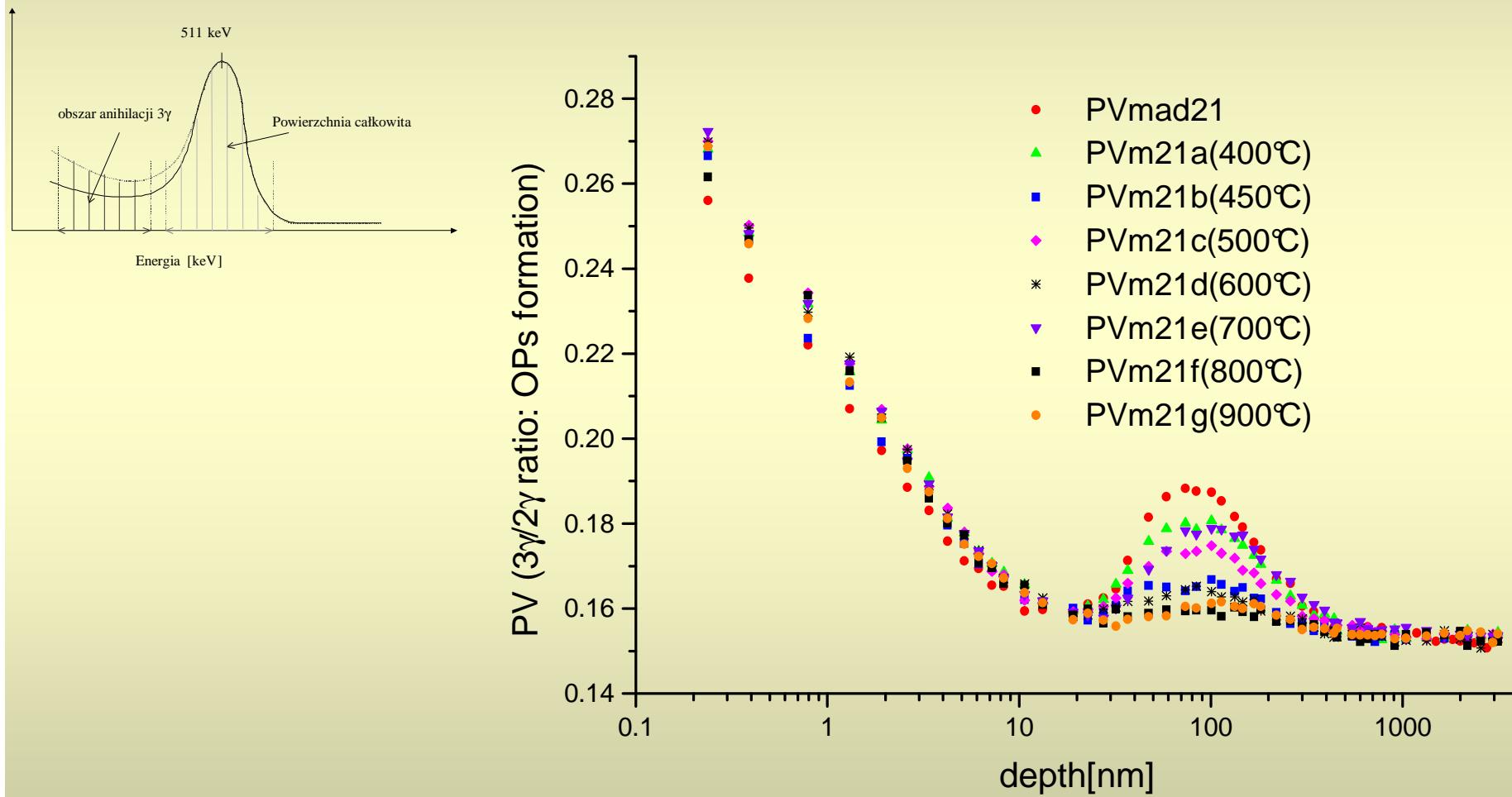


From K.Maex et al. J. Appl. Phys. 11, 93, 8793

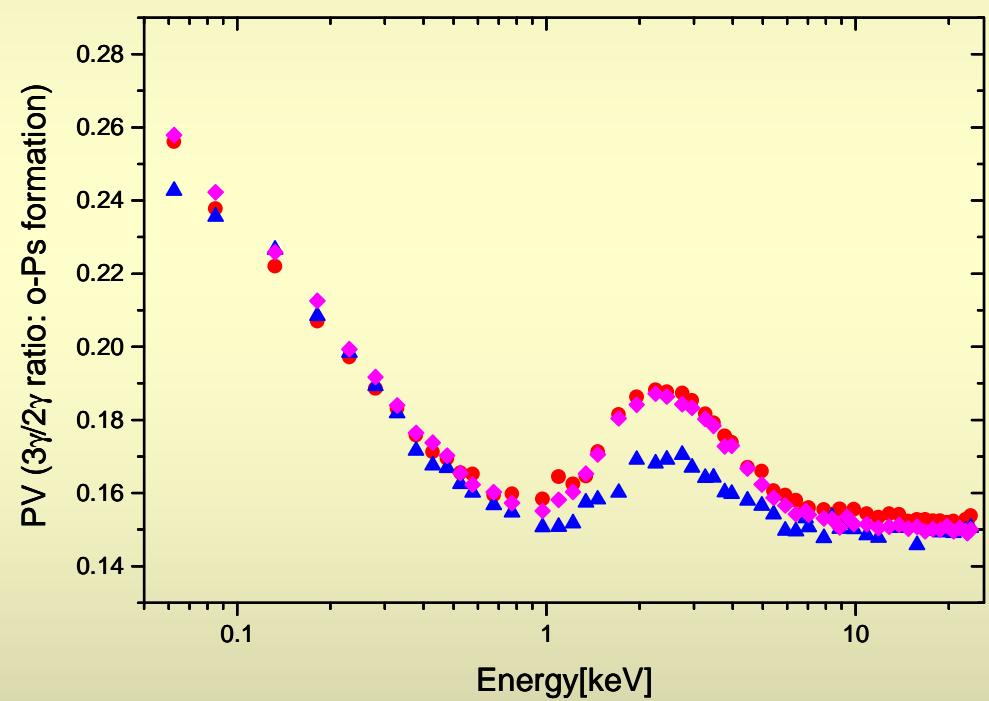
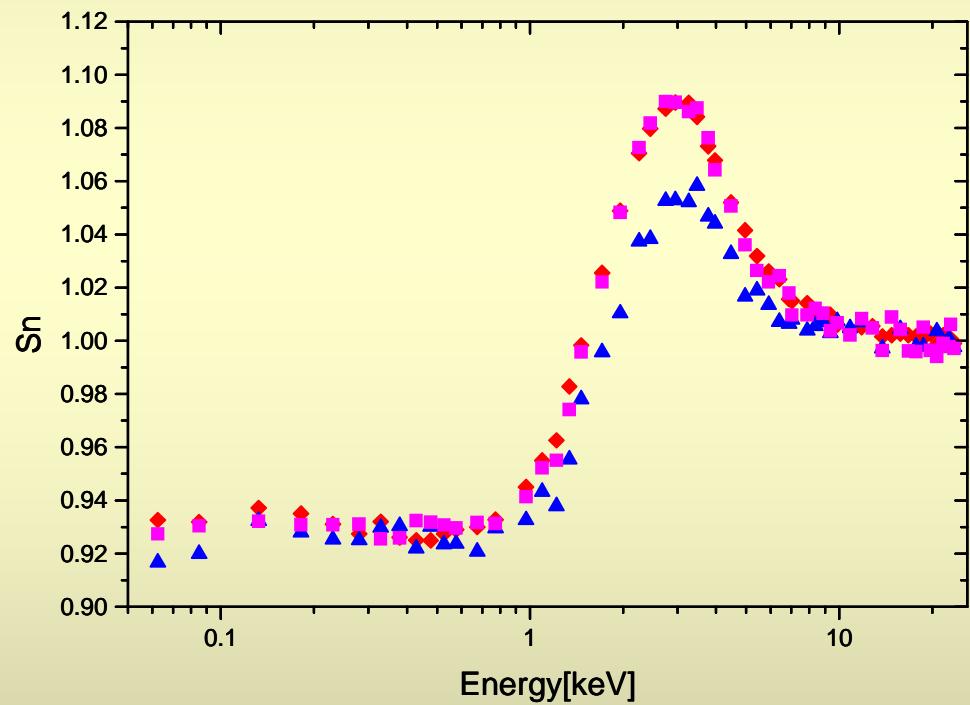
low ϵ materials - annealing



low ϵ materials - annealing

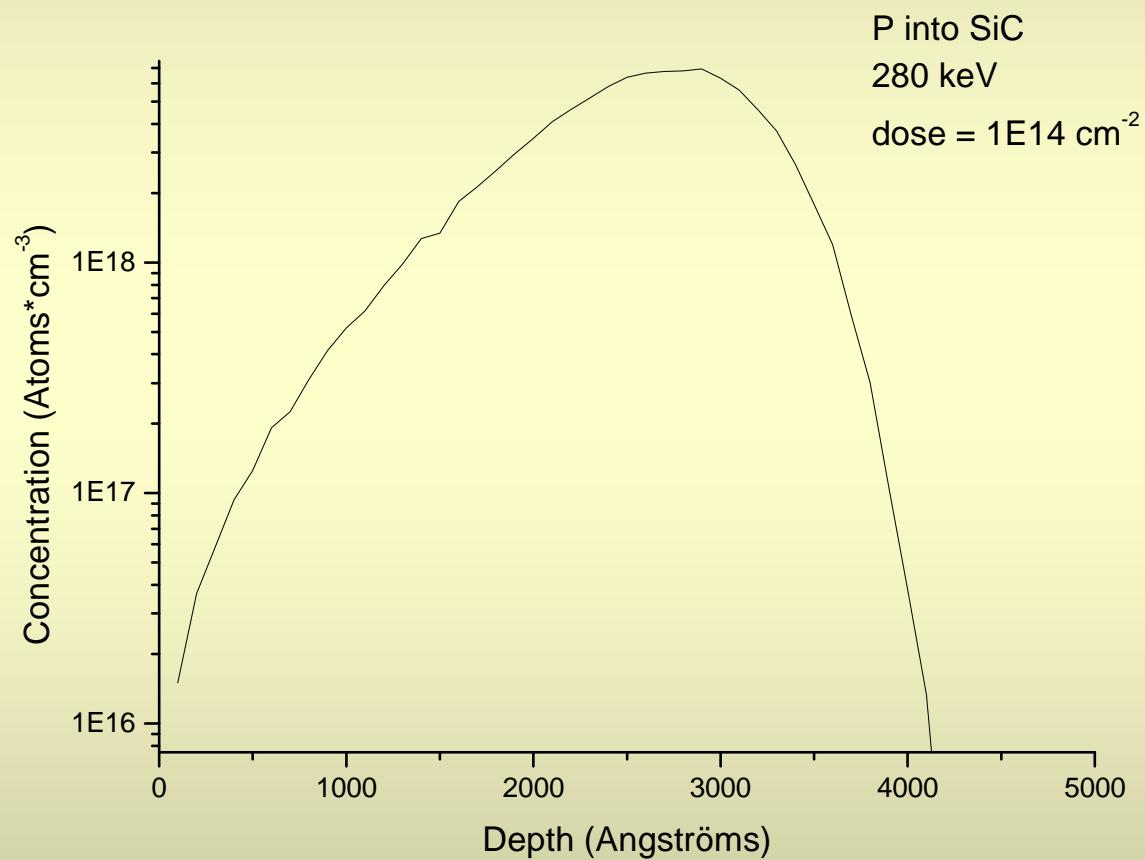


low ϵ materials - ageing



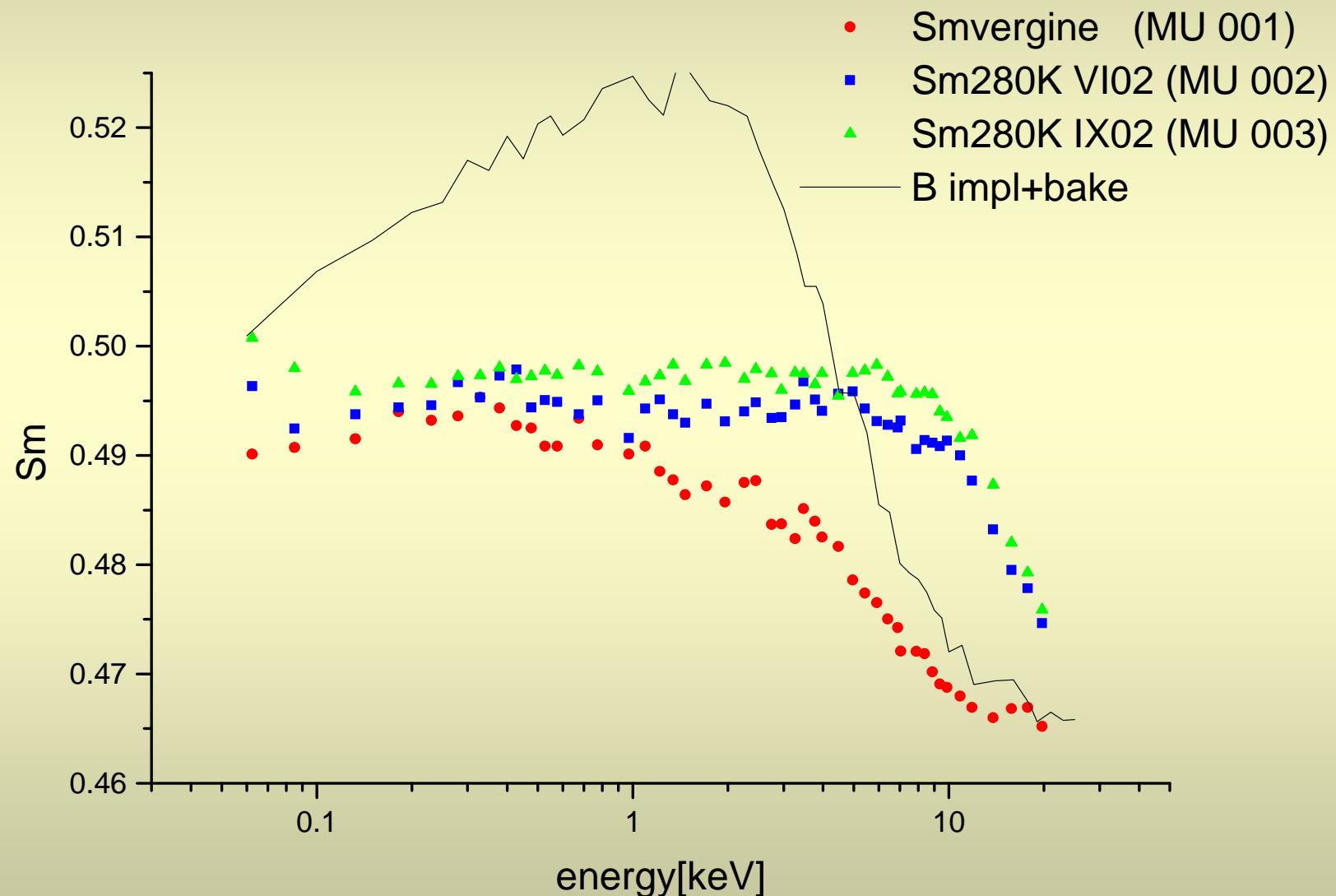
6H- SiC

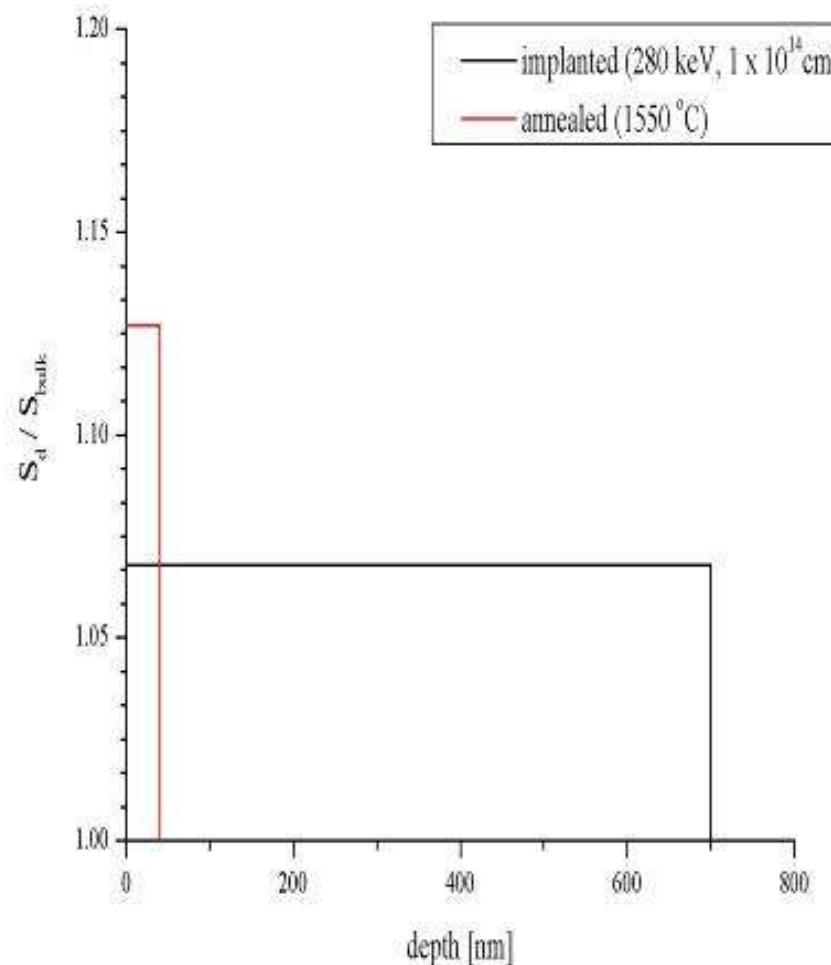
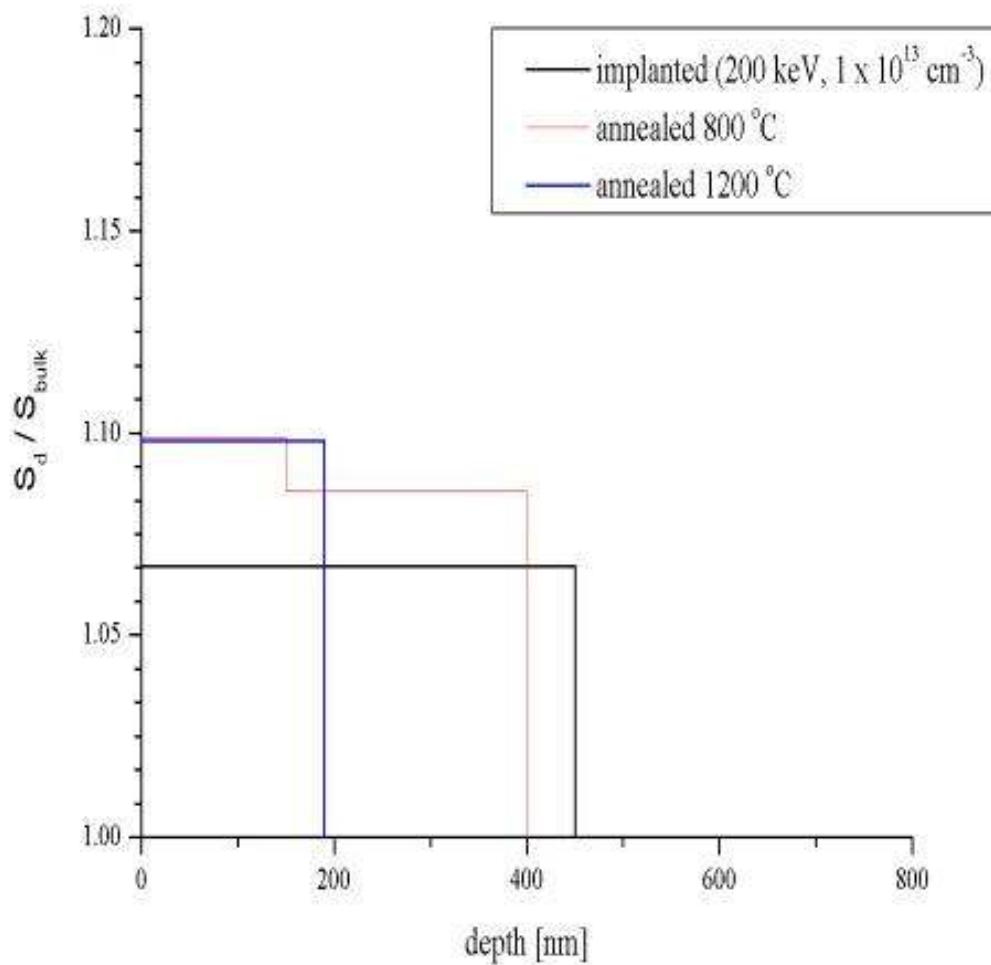
P⁺ implanted 10¹⁴cm⁻² dose @280 keV



Annealed in 1550°C for 30 min

“S”- parameter (= No. of defects)

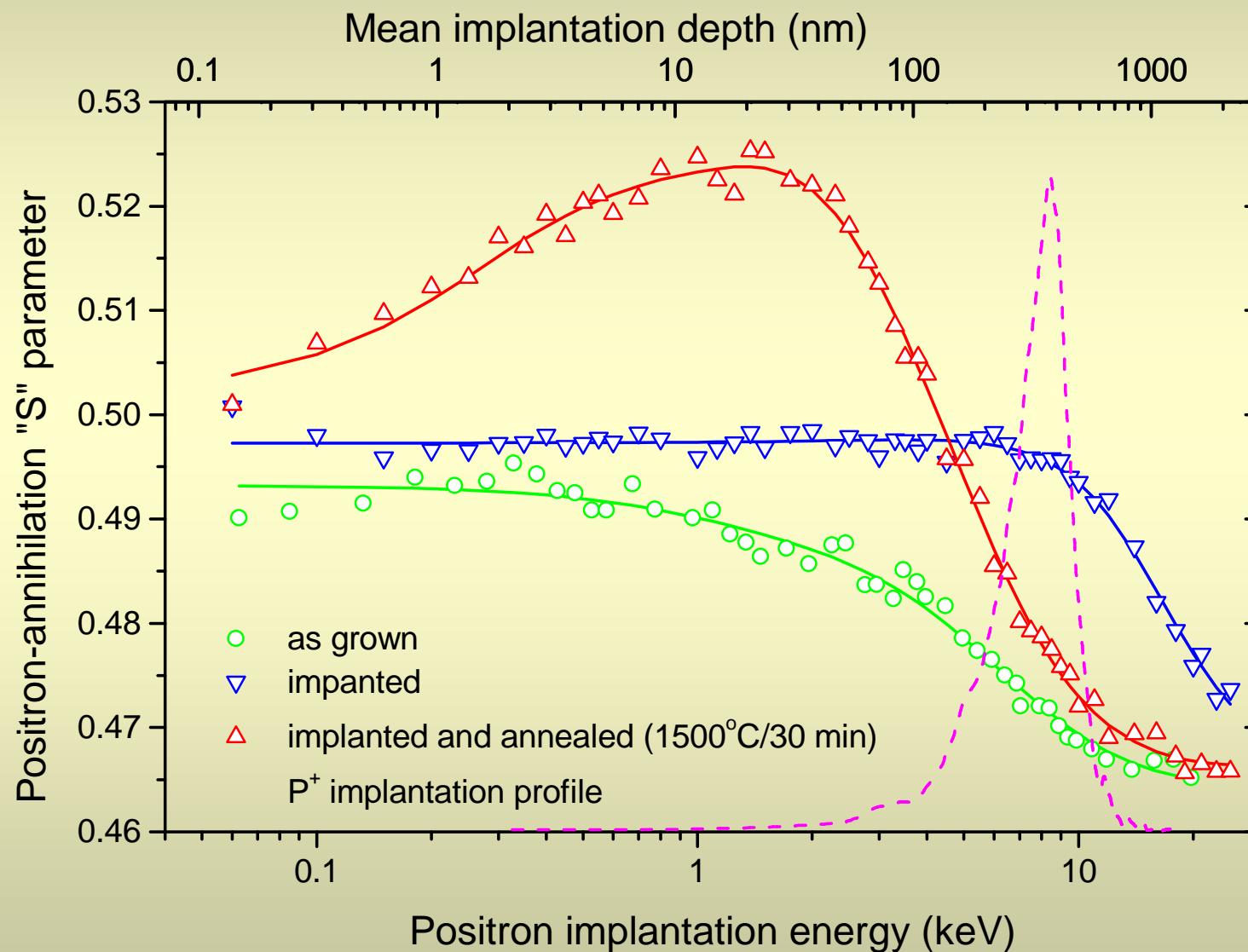




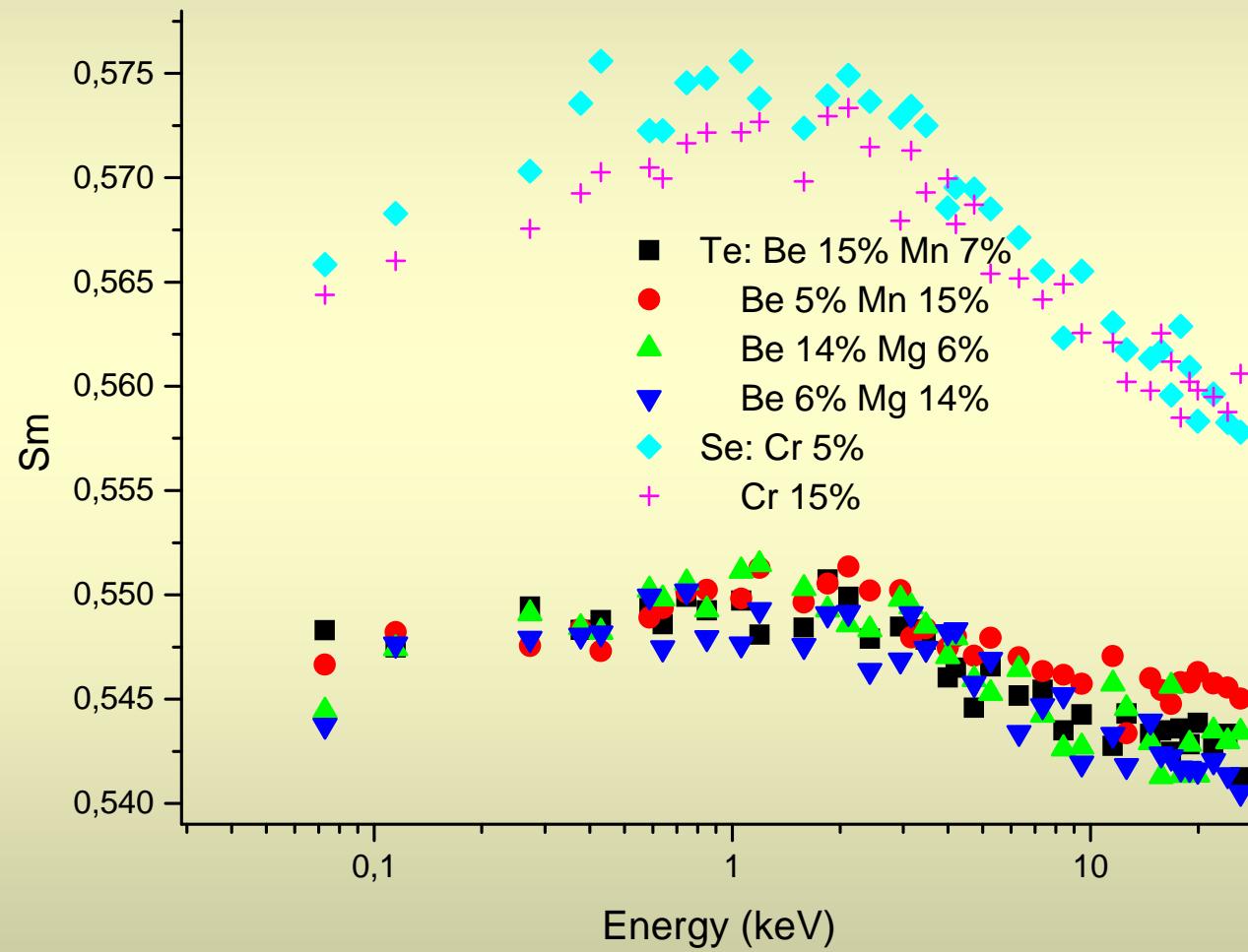
Ohshima *et al.*
[Appl Phys. A, **67**, 407-412 (1998)]

present

Modelling (VEPFIT)



ZnSe and ZnTe

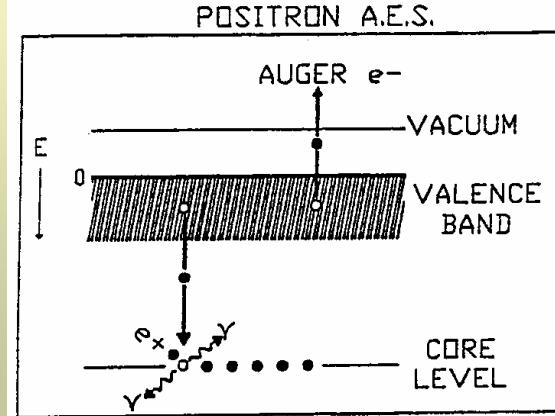
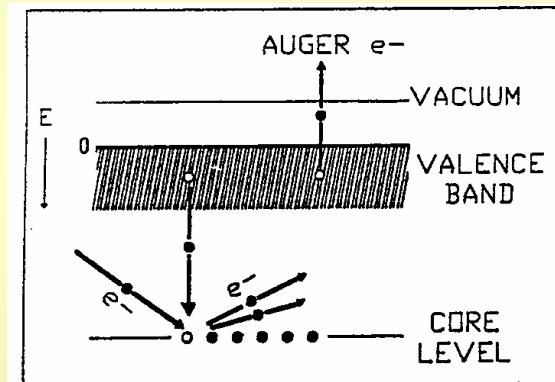


Positron Spectroscopy in Solid State Physics

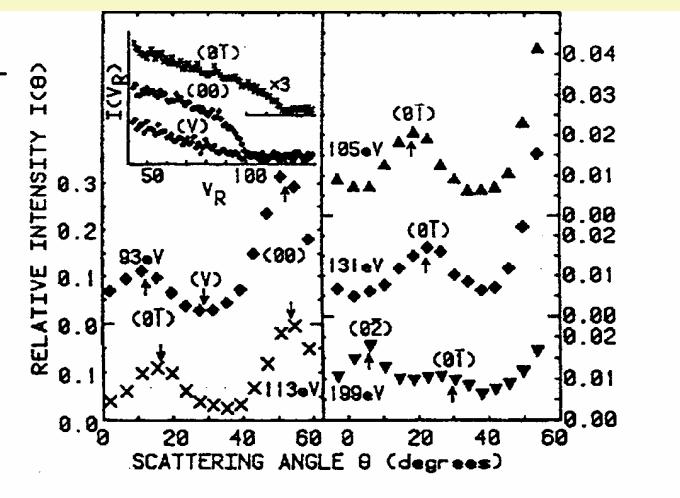
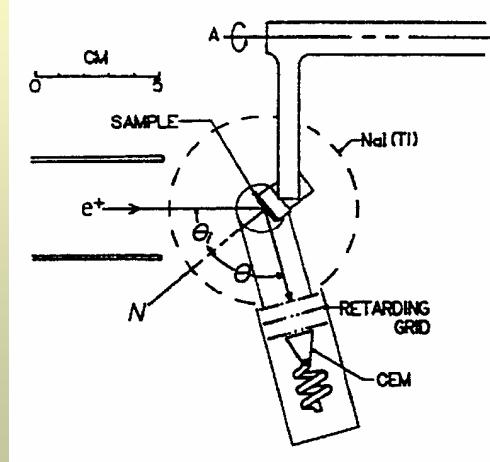
Future ?

Combined techniques!

Auger Spectroscopy



Low-energy Positron Diffraction



Genetic algorithm

Vol. 113 (2008)

ACTA PHYSICA POLONICA A

No. 5

Proceedings of the 37th Polish Seminar on Positron Annihilation, Lądek-Zdrój 2007

Genetic Algorithms for Positron Lifetime Data

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Recently, genetic algorithms have been applied for ultrafast optical spectrometry in systems with several convoluted lifetimes. We apply these algorithms and compare the results with POSFIT (by Kirkegaard and Eldrup) and LT programme (by Kansy). The analysis was applied to three types of samples: molybdenum monocrystals, Czochralski-grown silicon with oxygen precipitates, Si with under-surface cavities obtained by He + H ion co-implantation. In all three tests, the genetic algorithm performs very well, in particular for short lifetimes. Further developments to model the resolution function in genetic algorithms are needed.

Pozytony - narzędzie badania strukturalnych

Podziękowania:

Dr Antonio Zecca

Dr Roberto Brusa

Mgr Andrzej Karbowski

Prof. J. Fisz

Prof.. F. Firszt

Dziękuję!

More precipitates (in Trento)

